



# Amorphous core/shell Ti-doped SnO<sub>2</sub> with synergistically improved N<sub>2</sub> adsorption/activation and electrical conductivity for electrochemical N<sub>2</sub> reduction

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## ARTICLE INFO

### Article history:

Received 30 September 2021

Revised 17 December 2021

Accepted 22 December 2021

Available online 26 December 2021

### Keywords:

Nitrogen reduction reaction

Electrocatalysts

Density functional theory

Heteroatom doping

Synergistical effect

## ABSTRACT

Electrochemical nitrogen reduction reaction (NRR) has been considered as an appealing and sustainable method to produce ammonia from N<sub>2</sub> under ambient conditions, attracting increasing interest. Limited by low solubility of N<sub>2</sub> in water and high stability of N≡N triple bond, developing NRR electrocatalysts with both strong N<sub>2</sub> adsorption/activation and high electrical conductivity remain challenging. Here, we demonstrate an efficient strategy to develop NRR electrocatalyst with synergistically enhanced N<sub>2</sub> adsorption/activation and electrical conductivity by heteroatom doping. Combining computational and experimental study, the DFT-designed Ti-doped SnO<sub>2</sub> exhibits significantly enhanced NRR performance with ammonia yield rate of 13.09 μg h<sup>-1</sup> mg<sup>-1</sup> at -0.2 V vs. RHE. Particularly, the Faradaic efficiency reaches up to 42.6%, outperforming most of Sn-based electrocatalysts. The fundamental mechanism for improving NRR performance of SnO<sub>2</sub> by Ti doping is also revealed. Our work highlights a powerful strategy for developing high-activity electrocatalysts for NRR and beyond.

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As one of the most important industrial feedstock, ammonia (NH<sub>3</sub>) plays a vital role in modern society and has wide applications in chemical fertilizer, textile, pharmaceuticals, and related fields [1,2]. Also, it is considered as an efficient carbon-free energy carrier. Nowadays, ammonia production is still heavily dependent on traditional Haber-Bosch process with iron-based catalyst, in which high temperature and pressure are required to meet the harsh reaction conditions, inevitably producing huge amounts of greenhouse gas CO<sub>2</sub> emission. Therefore, it is essential to find out an alternative way for ammonia synthesis under ambient conditions that can remarkably reduce energy consumption and environmental pollution.

The conversion of N<sub>2</sub> to ammonia via electrochemical nitrogen reduction reaction (NRR) is a promising and sustainable strategy to produce ammonia under ambient conditions [3,4]. Recently, a number of electrocatalysts for NRR have been developed, including metal-free materials [5,6], noble metal-based materials [7,8], and transition metal-based materials [9–13]. However, the efficiency and space-time yield of NRR is still limited due to the low solubility in water and high bond energy of N≡N triple bond [14]. To this end, the potential electrocatalysts should enrich active sites to

stably adsorb N<sub>2</sub> molecule and efficiently activate inert N≡N triple bond [11,12]. Providing abundant active sites is regarded as the potential way to increase the apparent activity of NRR electrocatalysts [15]. Nanostructuring the catalyst, dispersing the catalyst on supports, and constructing single-atom catalyst could be the efficient strategies to achieve this. Then, tuning the intrinsic activity of each active site is the second key point to boost NRR performance of the electrocatalysts [16]. In general, the electronic structure of NRR catalyst significantly determines the intrinsic activity due to the proton-coupled electron transfer between N<sub>2</sub> molecule and the active site during NRR process [17]. Tailoring the electronic structure of a catalyst could directly change the adsorption behavior of N<sub>2</sub> molecule and activate N≡N triple bond to facilitate the conversion from N<sub>2</sub> molecule to ammonia.

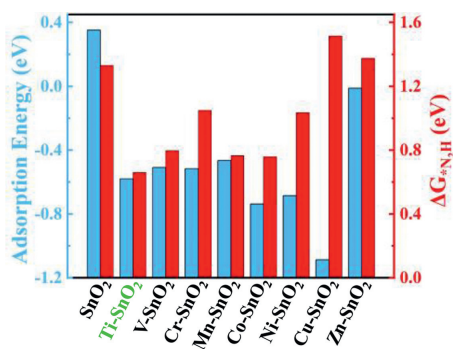
Heteroatom doping can tailor the structure of energy band and tune the electron transfer behavior of the semiconductors to overcome the sluggish reaction kinetics, boosting the catalytic activity. Some works demonstrated that the doped heteroatom could act as the catalytic site and redistribute the charge location on the surface of electrocatalyst, enhancing the chemical adsorption for N<sub>2</sub> molecule [18–20]. The generated unsaturated coordination, vacancy and tunable electronic structure could activate the adsorbed N<sub>2</sub> molecule to boost NRR performance [21–25].

As a semiconductor material, SnO<sub>2</sub> has high stability and low hydrogen evolution reaction (HER) activity, being potential cata-

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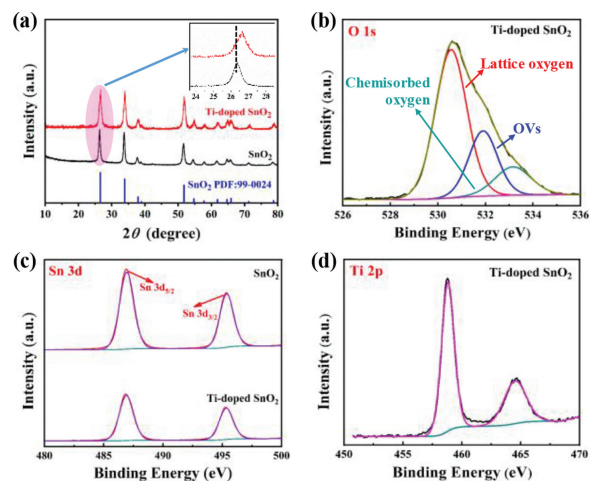
**Fig. 1.** The calculated  $N_2$  adsorption energies (in blue) and the Gibbs free energy changes of the first hydrogenation step ( $\Delta G^*_{N_2H}$ , in red) of transition metal-doped  $SnO_2$ .

lyst for electrochemical NRR. It suffers from low electrochemical NRR performance owing to poor adsorption for  $N_2$  molecule, limited catalytic sites, and low electrical conductivity [26]. We propose that transition metal doping could boost the NRR performance of  $SnO_2$ . Transition metals have diverse d-orbital electronic structures and could promote electron transfer with  $N_2$  molecule via acceptance-donation effect [27]. The strong interactions could decrease the bond order of  $N_2$  molecule and weaken  $N\equiv N$  triple bond, facilitating the adsorption/activation of  $N_2$  molecule. Also, transition metals generally have relatively high electrical conductivity compared to  $SnO_2$ , favorable for electron transfer.

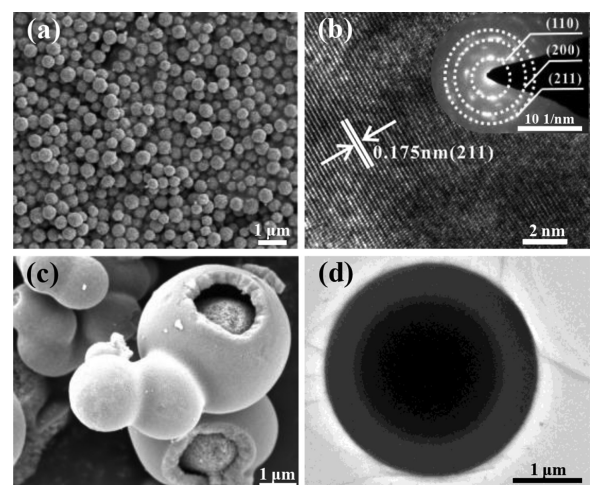
Herein, we demonstrated this strategy by doping transition metal into  $SnO_2$  to boost the electrochemical NRR performance. Based on DFT screening, Ti-doped  $SnO_2$  has been proven experimentally to exhibit significantly improved electrochemical NRR performance with yield rate of  $13.09 \mu g h^{-1} mg^{-1}$  and Faradaic efficiency of 42.6% at  $-0.2 V$  vs. RHE.

A series of transition metals, including Ti, V, Cr, Mn, Co, Ni, Cu and Zn, were chosen as the heteroatoms to construct transition metal-doped  $SnO_2$  catalysts. To predict their electrochemical NRR activity, DFT calculations were performed to quantitatively evaluate the  $N_2$  adsorption/activation and thermodynamically hydrogenation process. Generally, the adsorption interaction between  $N_2$  molecule and the electrocatalyst should be not too strong or not too weak, simultaneously facilitating the adsorption of  $N_2$  molecule and the desorption of the final product. DFT results showed that  $N_2$  molecule can be stably adsorbed on transition metal atoms with end-on configuration, rather than on Sn or O atoms. As shown in Fig. 1 and Table S1 (Supporting information), one can see that Ti-, V-, Cr- and Mn-doped  $SnO_2$  have much lower adsorption energies than  $SnO_2$ , but much higher than others excluding Zn-doped  $SnO_2$ , exhibiting the moderate adsorption interaction. Starting from end-on configuration,  $N_2$  molecule could undergo the distal/alternating pathway to successively produce two ammonia molecules *via* continuous hydrogenation steps and the first one to form  $*N_2H$  ( $*N_2 \rightarrow *N_2H$ ) is generally considered as the potential-determining step (PDS) to determine the NRR activity of catalysts [28,29]. Therefore, we calculated the Gibbs free energy change of this step for  $SnO_2$  and all as-designed electrocatalysts. Among them, Ti-doped  $SnO_2$  has the lowest  $\Delta G^*_{N_2H}$  with the value of 0.66 eV in Fig. 1. It is also lower than the previously reported NRR electrocatalysts with high activity, such as  $MoS_2$  with 0.68 eV [30] and Ru single-atom catalyst with 0.73 eV [31]. Therefore, Ti-doped  $SnO_2$  is predicted the potential NRR electrocatalyst that need to be proven by experimental study.

$SnO_2$  and Ti-doped  $SnO_2$  were synthesized by a facile solvothermal method and no nitrogen-containing raw materials were involved to avoid ammonia contamination as much as possible. As

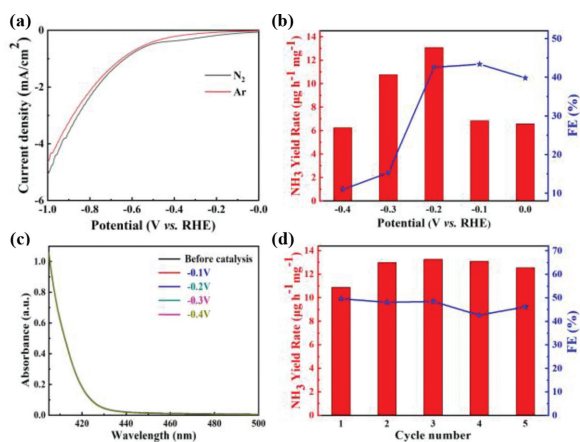


**Fig. 2.** (a) XRD patterns of  $SnO_2$  and Ti-doped  $SnO_2$  and XPS spectra for (b) O 1s, (c) Sn 3d and (d) Ti 2p.



**Fig. 3.** SEM images and HRTEM images of (a, b)  $SnO_2$  and (c, d) Ti-doped  $SnO_2$ .

shown in X-ray diffraction (XRD) patterns in Fig. 2a, one can see that the characteristic diffraction peaks of  $SnO_2$  locate at  $26.58^\circ$ ,  $33.86^\circ$  and  $51.76^\circ$ , corresponding to the (110), (101) and (211) planes of  $SnO_2$  (JCPDS No. 99-0024), respectively. The peaks of Ti-doped  $SnO_2$  well correspond to  $SnO_2$  without any impurity or alien phase, showing that Ti is incorporated into  $SnO_2$  [32–34]. The inset exhibits a slight shift of the peak position to high angle, meaning a solid solution with the doped Ti element [35]. The surface chemical states of the as-designed catalysts were characterized to evidence the presence of O, Sn and Ti elements by X-ray photoelectron spectroscopy (XPS) as shown in Figs. 2b–d. The peaks at 458.82 and 464.54 eV in the region of Ti 2p are ascribed to  $Ti^{4+}$  [36]. The typical peaks for Sn  $3d_{5/2}$  and  $3d_{3/2}$  at 486.9 and 495.3 eV indicate that Sn element mainly exists as  $Sn^{4+}$  in Ti-doped  $SnO_2$  [37]. The peak positions have ignored shift but the intensities are significantly different in comparison to those in  $SnO_2$  and this could be caused by Ti doping [38]. For the O 1s region, the peaks at 530.43, 531.84 and 533.54 eV are ascribed to lattice oxygen, oxygen atoms in vicinity of the oxygen vacancy and chemisorbed oxygen, respectively [39]. Scanning electron microscopy (SEM) image and the high-resolution transmission electron microscopy (HRTEM) image in Fig. 3 show that  $SnO_2$  has uniform hexagonal nanosphere morphology and (211) crystal plane is exposed with an interplanar distance of 0.175 nm. In comparison, Ti-doped  $SnO_2$  clearly shows an amorphous hollow core/shell

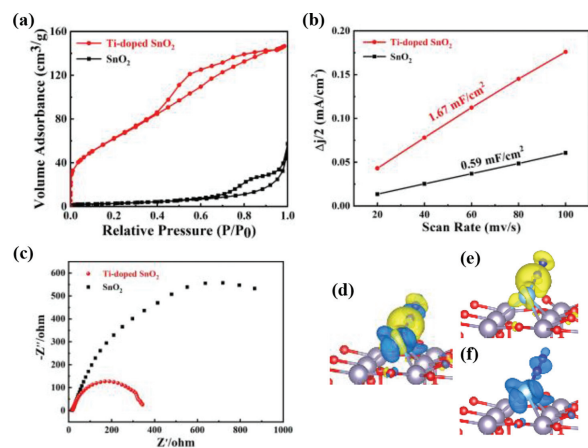


**Fig. 4.** The electrochemical NRR performance of Ti-doped SnO<sub>2</sub>. (a) LSV curves in Ar- and N<sub>2</sub>-saturated 0.1 mol/L Na<sub>2</sub>SO<sub>4</sub>. (b) NH<sub>3</sub> yield rates and Faradaic efficiency at different applied potentials. (c) Chronoamperometry curves for N<sub>2</sub>H<sub>4</sub> at different applied potentials. (d) Recycling tests at -0.2 V vs. RHE.

structure with holey shell [38,40]. SAED result also supports amorphous structure shown in Fig. S1 (Supporting information). Interestingly, Ti doping remarkably increases the size of SnO<sub>2</sub> particles. This could be caused by the formation of core/shell Ti-doped SnO<sub>2</sub> via Ostwald ripening [38].

The standard electrochemical NRR experiments were performed in a two-compartment cell separated by a Nafion 117 membrane in 0.1 mol/L Na<sub>2</sub>SO<sub>4</sub> electrolyte under ambient conditions. During the measurement, N<sub>2</sub> gas was continually supplied to the working electrode to react with H<sup>+</sup> from the electrolyte and electron from the electrode react to produce NH<sub>3</sub> on the surface of the working electrode. The produced NH<sub>3</sub> and possible by-product N<sub>2</sub>H<sub>4</sub> were quantitatively determined using the indophenol blue method [41] and Watt-Chrisp method [42] with the corresponding calibration curves in Figs. S2 and S3 (Supporting information), respectively. In Fig. 4a, the linear sweep voltammetry (LSV) curves show that Ti-doped SnO<sub>2</sub> has larger current density in N<sub>2</sub>-saturated electrolyte than in Ar-saturated condition, corresponding to N<sub>2</sub> response for electrochemical NRR. The ammonia yield rates and Faradaic efficiency of Ti-doped SnO<sub>2</sub> at various electrode potentials on a RHE scale are shown in Fig. 4b with the maximum values of 13.09 μg h<sup>-1</sup> mg<sup>-1</sup> and 42.6% at -0.2 V, respectively. The possible by-product N<sub>2</sub>H<sub>4</sub> was not detected, exhibiting the excellent selectivity of Ti-doped SnO<sub>2</sub> for converting N<sub>2</sub> to ammonia in Fig. 4c. Also, it shows a prominent cycling stability in 5 cycles shown in Fig. 4d. In comparison, SnO<sub>2</sub> has the ammonia yield rate of 8.23 μg h<sup>-1</sup> mg<sup>-1</sup> at -0.8 V with Faradaic efficiency of 5.6% (Fig. S5 in Supporting information) under the similar conditions, consistent with the previous report of cubic sub-micron SnO<sub>2</sub> particles [26]. Evidently, Ti doping can significantly improve the electrochemical NRR performance of SnO<sub>2</sub>, being an efficient strategy for developing a high-activity electrocatalyst. Moreover, such excellent NRR performance of Ti-doped SnO<sub>2</sub>, particularly for Faradaic efficiency, surpasses most reported Sn-based electrocatalysts for electrochemical NRR under ambient conditions listed in Table S2 (Supporting information) [11,26,31,43–46].

To uncover the origin of the produced NH<sub>3</sub> during electrochemical NRR process, a series of control experiments were constructed [47]. As shown in Fig. S6 (Supporting information), the produced ammonia in each experiment can be reasonably ignored in comparison to that in the standard electrochemical NRR test. This demonstrates that ammonia can only be electrochemically synthesized on the surface of Ti-doped SnO<sub>2</sub> from the supplied N<sub>2</sub> gas. In addition, no nitrogen-containing raw materials were used



**Fig. 5.** (a) Nitrogen sorption isotherm, (b) the electrochemical double-layer capacitance analysis at -0.25 V, and (c) EIS analysis of SnO<sub>2</sub> and Ti-doped SnO<sub>2</sub>. (d) Difference charge density of N<sub>2</sub> adsorbed on Ti-doped SnO<sub>2</sub> with positive (e) and negative (f) charges represented in yellow and blue, respectively.

to synthesize the catalysts mentioned above, implying that the produced ammonia could not be from other nitrogen-containing compounds. In order to exclude the contamination from the environment, the measured ammonia in Ar condition could be subtracted [47]. Therefore, ammonia yield rate was calculated to be 12.35 μg h<sup>-1</sup> mg<sup>-1</sup>.

The mechanism of the boosted electrochemical NRR performance of SnO<sub>2</sub> by Ti doping is very informative for understanding the intrinsic catalytic activity of Ti-doped SnO<sub>2</sub> and developing high-activity electrocatalysts for NRR. The good catalysts could have relatively large specific surface area to expose more active sites. The BET results in Fig. 5a exhibit 10 fold increase of the specific surface area of Ti-doped SnO<sub>2</sub> with 230.9 m<sup>2</sup>/g compared to SnO<sub>2</sub> with 24.3 m<sup>2</sup>/g. It could be caused by the morphology change to generate an amorphous hollow core/shell structure with holey shell due to Ti doping. Furthermore, the electrochemical double-layer capacitance (C<sub>dl</sub>) measurements in Fig. 5b show that Ti-doped SnO<sub>2</sub> has much bigger electrochemically active surface areas than SnO<sub>2</sub>, conducive to the improved electrochemical NRR performance. Faradaic efficiency for NRR is directly related with the electrical conductivity of electrocatalysts. The EIS analysis in Fig. 5c shows that electrical conductivity of SnO<sub>2</sub> is significantly improved by Ti doping, contributing to the markedly increase of FE, supporting the feasibility of our design strategy.

DFT calculations were performed to further investigate the role for Ti doping in improving the catalytic performance of SnO<sub>2</sub>. Firstly, Gibbs free energy diagrams of NRR on SnO<sub>2</sub> and Ti-doped SnO<sub>2</sub> (Fig. S7 in Supporting information) were calculated to reveal the thermodynamic behavior in detail. In both distal pathway and alternating pathway, the first dehydrogenation step is determined to be PDS with the biggest energy change, proving the accuracy of our prediction mentioned above. The doping of Ti atom contributes the stability of \*NNH intermediate. Secondly, the charge difference density in Figs. 5d–f illustrates that there is electron accumulation and consumption on both Ti atom and N<sub>2</sub> when N<sub>2</sub> is adsorbed on Ti-doped SnO<sub>2</sub>. This is a two-way charge transfer between Ti atom and N<sub>2</sub> to achieve acceptance-donation process, favorable for N<sub>2</sub> activation on the catalyst. It is also consistent with the atomic charge variations at each step along the entire distal pathway (Fig. S8 in Supporting information) that both Ti and SnO<sub>2</sub> can accept the electron from the adsorbed N<sub>2</sub> molecule, carrying on the negative charges when N<sub>2</sub> molecule is adsorbed on Ti-doped SnO<sub>2</sub>. During the whole NRR process, Ti atom acts as the emissary for electron transport between the adsorbed state \*N<sub>x</sub>H<sub>y</sub> and SnO<sub>2</sub> substrate.

Thirdly, according to the partial density of states (PDOS) in Fig. S9 (Supporting information), Ti atom has a strong  $d-\pi^*$  orbital coupling with  $N_2$  around the Fermi level. There is an overlap between  $d$ -orbital of Ti atom and molecular orbital of the adsorbed  $N_2$  that causes the "push-pull" effect [11], facilitating  $N_2$  activation. This is also in well agreement with the analysis of the charge difference density discussed above.

In summary, we demonstrated a heteroatom doping strategy to synergistically enhance  $N_2$  adsorption/activation and electrical conductivity, boosting the electrochemical NRR activity of semiconductor material. Seven common transition metals were selected as the heteroatom to be doped in  $SnO_2$ , as they generally have diverse  $d$ -orbital electronic structures and much higher electrical conductivity. Based on DFT prediction on  $N_2$  adsorption and NRR kinetics, Ti-doped  $SnO_2$  was found to be the most potential electrocatalyst for NRR. Then, a facile solvothermal method was used to prepare Ti-doped  $SnO_2$ . The obtained Ti-doped  $SnO_2$  exhibits amorphous hollow core/shell structure with holey shell and much higher activity for electrochemical NRR than  $SnO_2$  with ammonia yield rate and Faradaic efficiency of  $13.09 \mu\text{g h}^{-1} \text{mg}^{-1}$  and 42.6% at  $-0.2 \text{ V}$ , respectively. Ti doping simultaneously causes multiple effects on  $SnO_2$ , including morphology, electrochemical characters, electronic structures, and charge distribution, which facilitate the  $N_2$  adsorption/activation. The improved electrical conductivity promotes the electron transfer during electrochemical hydrogenation process, conducive to Faradaic efficiency.

#### Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

#### Acknowledgment

This work was supported by the National Natural Science Foundation of China (No. U2067216). The authors acknowledge Beijing PARATERA Tech Co., Ltd. for providing HPC resources that have contributed to the research results reported within this paper (URL: <https://paratera.com/>).

#### Supplementary materials

Supplementary material associated with this article can be found, in the online version, at doi:10.1016/j.ccl.2021.12.054.

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