



## Communication

Two-step colloidal synthesis of micron-scale Bi<sub>2</sub>O<sub>2</sub>Se nanosheets and their electrostatic assembly for thin-film photodetectors with fast responseXixi Pang<sup>a,b</sup>, Yingtao Zhao<sup>a,b</sup>, Xiangxiang Gao<sup>a,b</sup>, Guoqi Wang<sup>a,b</sup>, Hao Sun<sup>a,b</sup>, Jun Yin<sup>a,b,\*</sup>, Jian Zhu<sup>a,b,c,\*</sup><sup>a</sup> School of Materials Science and Engineering, Nankai University, Tianjin 300350, China<sup>b</sup> National Institute for Advanced Materials, Nankai University, Tianjin 300350, China<sup>c</sup> Tianjin Key Laboratory for Rare Earth Materials and Applications, Nankai University, Tianjin 300350, China

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## ABSTRACT

Recently discovered bismuth oxychalcogenide (Bi<sub>2</sub>O<sub>2</sub>Se) has aroused great interest due to its ultrahigh carrier mobility, tunable band gap and good environmental stability, making it a promising candidate for high-performance electronics and optoelectronics. Their synthesis by colloidal approaches represents a cost-effective alternative to well-established chemical vapor deposition methods, and the resulting electronic-grade inks are important for large-area printed or wearable electronics. However, it is still challenging to control the colloidal growth of Bi<sub>2</sub>O<sub>2</sub>Se nanosheets in solution in addition to their assembly into high-performance thin films. Here, we report a two-step colloidal synthesis of Bi<sub>2</sub>O<sub>2</sub>Se nanosheets by separating the seeding and growth steps, thereby achieving controllable production of nanosheets with a lateral size of 1.4 μm and a thickness of 10 nm at optimized reaction conditions. These Bi<sub>2</sub>O<sub>2</sub>Se nanosheets are electrostatically assembled into large-area thin films, from which a photodetector is fabricated with a responsivity of 6.1 A/W and a short response time of 368 μs under the 520-nm laser illumination. The device exhibits fast response to modulations as high as 100 kHz, along with a -3 dB bandwidth of 1 kHz. This work provides an important understanding of the controlled colloidal synthesis of Bi<sub>2</sub>O<sub>2</sub>Se nanosheets, and demonstrates their potential applications in fast photodetectors.

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Two-dimensional (2D) materials are competitive candidates for next-generation electronic and optoelectronic devices. Their flat surfaces free of dangling bonds, atomically thin thicknesses, and high carrier mobilities bring merits of efficient channel current regulation and a high degree of vertical integration of different types of 2D materials [1–5]. The past decade has witnessed an extraordinary increase in the family of 2D materials including graphene [6–8], black phosphorus (BP) [9–11], and transition metal dichalcogenides (TMDs) [12,13]. However, in these 2D materials, graphene is difficult to be switched off [14,15], TMDs generally have low carrier mobilities [3,16,17], and BP is not air-stable [18,19]. Thus, it is urgent to find new type of 2D materials with high carrier mobilities, suitable band gaps, and good ambient stability

simultaneously. Bi<sub>2</sub>O<sub>2</sub>Se has recently emerged as a bismuth-based oxychalcogenide 2D material, which shows a tetragonal structure with the *I4/mmm* space group. It consists of planar covalently bonded oxide layers (Bi<sub>2</sub>O<sub>2</sub>) weakly connected by Se square arrays [20]. With the advantages of high carrier mobility (>20,000 cm<sup>2</sup> V<sup>-1</sup> s<sup>-1</sup> at 2 K), layer-number tunable band gaps, low subthreshold swing (SS, ~65 mV/dec) [21,22] and air stability, it is particularly suitable for high-performance transistors or photodetectors.

While bulk Bi<sub>2</sub>O<sub>2</sub>Se was synthesized by solid-state reaction as early as 1973 [23], the recent syntheses of Bi<sub>2</sub>O<sub>2</sub>Se ultrathin films with controlled thicknesses, good crystallinity and large sizes via chemical vapor deposition (CVD) have spurred the fundamental studies of their properties and applications [21]. However, CVD approaches typically suffer from the disadvantages of low product yield, high equipment investment, as well as the complex and strict technical processes. In addition, complicated transfer steps are necessary if different substrates are needed for device fabrications,

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**Fig. 1.** Schematic diagram of a two-step synthesis process of the  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets.

which makes it difficult for large-scale integration. The wet chemistry synthesis of  $\text{Bi}_2\text{O}_2\text{Se}$  has emerged as an alternative mass-production approach with benefits of mild conditions, controllability, inexpensiveness, and high yield. For example,  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets with thicknesses of 155 nm and clear edges have been produced by a molten salt hydrothermal method [24];  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets with thicknesses of 2 nm and lateral sizes of 100–200 nm have been produced by reacting selenourea with bismuth nitrate in alkaline solutions at room temperature [25]. A recent work also illustrates the importance of organic ions on the thicknesses of  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets during hydrothermal reactions [26]. These methods are simple and feasible on a large scale, but the synthesized products typically have inhomogeneous distributions in thicknesses or sizes, and tend to agglomerate due to the absence of stabilizing ligands.

Here, we propose a new synthesis method to better control the phase and the morphology of  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets by separating the generation of seed crystals and the growth of nanosheets. We have systematically studied the influence of different solvents, additives and reaction conditions on the sizes and morphologies of  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets. The large and ultrathin nanosheets attained at optimized conditions are further assembled into thin films by electrostatic assembly. Using these thin films, we fabricate high performance photodetectors with responsivity of  $\sim 6.1$  A/W and response time of  $\sim 368$   $\mu\text{s}$ , which are superior to those prepared by other solution-processed  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets, or even those synthesized by CVD [26–31]. The preparation and assembly of

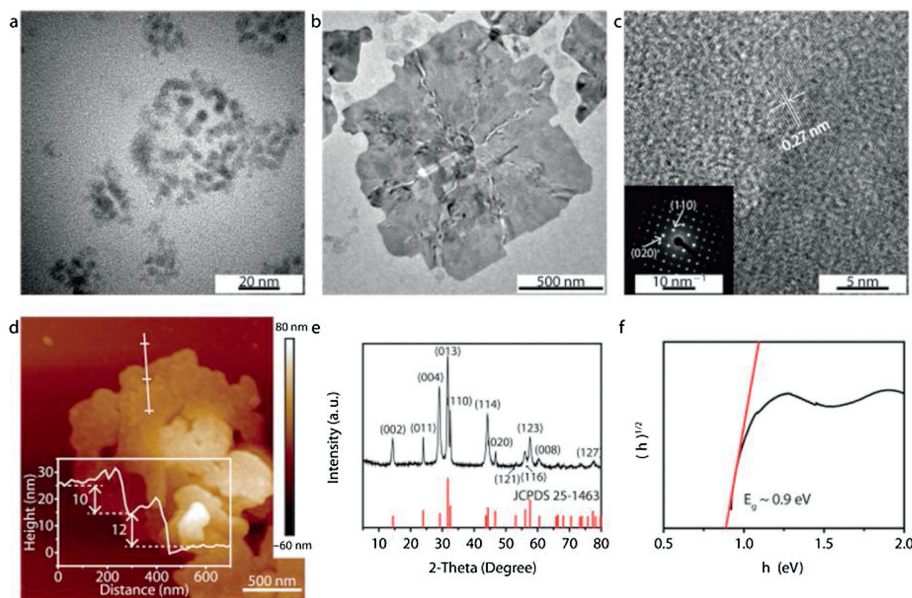
colloidal  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets are of great significance to the practical realization of thin film 2D materials for future electronic applications.

To better control the synthesis of  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets, the seeding and growing steps are separated in our procedure, which is schematically shown in Fig. 1. The overall reaction in the seeding step is shown as follows:

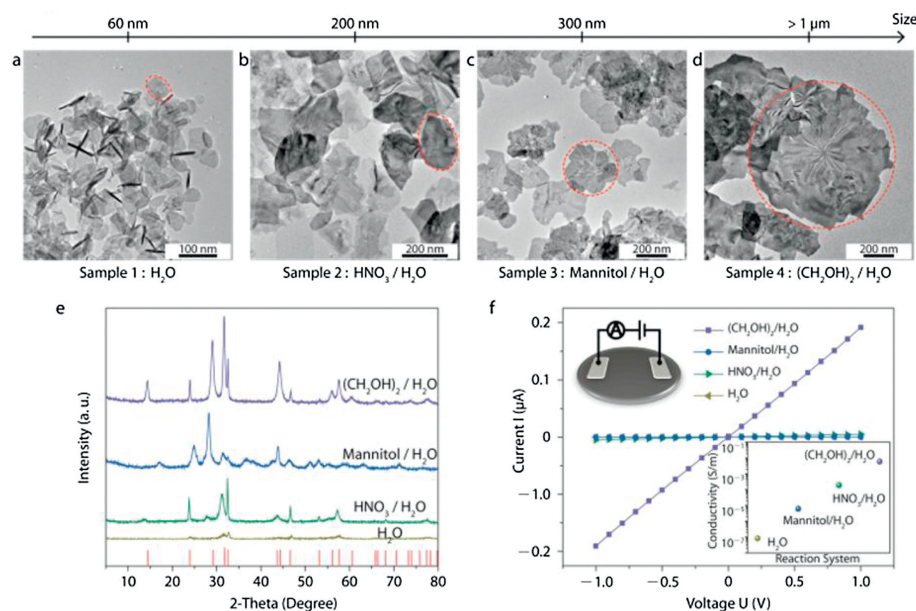


The citrate anions ( $[\text{C}_6\text{O}_7\text{H}_5]^{3-}$ ) released by the dissolution of sodium citrate are chelated with  $\text{Bi}^{3+}$  to produce  $[\text{C}_6\text{O}_7\text{H}_5]^{3-}\text{Bi}^{3+}$  complexes in solution 1 (Eq. 1) [32,33]. The selenide ions ( $\text{Se}^{2-}$ ) from the decomposition of selenourea ( $\text{SeC}(\text{NH}_2)_2$ ) in alkaline medium are produced along with cyanamide ( $\text{H}_2\text{NCN}$ ) in solution 2 (Eq. 2). When the bismuth nitrate solution is added dropwise to the selenourea solution,  $\text{Bi}^{3+}$  slowly released from the complexes react with  $\text{H}_2\text{O}$  and  $\text{Se}^{2-}$  to form  $\text{Bi}_2\text{O}_2\text{Se}$  seeds through a thermodynamically favorable process (Eq. 3). The as-synthesized seeds have a morphology of round disk with an average diameter of  $5.5 \pm 0.6$  nm by TEM (Fig. 2a).

In the subsequent growth step, the seeds undergo Ostwald ripening [34] in the hydrothermal conditions at  $150^\circ\text{C}$ , and grow laterally into  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets with a size as large as  $\sim 1.4$   $\mu\text{m}$  after 6 h (Fig. 2b). The  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets are further analyzed by high-resolution TEM and selected-area electron diffraction (SAED) to confirm the crystallinity. As shown in Fig. 2c, the high-resolution TEM image shows a lattice spacing of 0.27 nm, corresponding to the plane (110) of  $\text{Bi}_2\text{O}_2\text{Se}$ , and the strong SAED pattern (Fig. 2c inset) indicates the single crystalline nature of the as-synthesized nanosheets. The AFM measurements reveal an average thickness of 10 nm, indicating a multilayer structure (Fig. 2d). Furthermore, Fig. 2e depicts a typical XRD pattern congruent with tetragonal structure (JCPDS No. 25-1463). The diffraction peaks at  $32.253^\circ$  and



**Fig. 2.** Morphologies and characterizations of the synthesized  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets. TEM images of (a) crystal seeds after mixing Solution 1 and 2, and (b)  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets. (c) High-resolution TEM image of the  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets, along with an inset SAED pattern demonstrating its single crystalline nature. (d) AFM image of the multiple  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets and the height profile with a thickness of 10 nm. (e) XRD pattern of the synthesized  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets. (f) Relationship between the rescaled absorption factor  $(ah\nu)^{1/2}$  and the photon energy  $h\nu$ , showing a fitted optical band gap of 0.9 eV.



**Fig. 3.** Characterizations of the  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets by tuning the solvents. Morphology of the  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets obtained from the solvent of (a) water (Sample 1), (b)  $\text{HNO}_3$  solution (Sample 2), (c) mannitol solution (Sample 3), and (d) the mixture of ethylene glycol and water (Sample 4). (e) XRD patterns of the  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets in Samples 1–4. (f) Electrical conductivity of four compacted samples of  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets. Inset: A schematic diagram for measurement.

$46.061^\circ$  correspond to the crystal planes of (110) and (020) denoted in Fig. 2c. No impurity peaks are observed according to the standard card, which indicates that the crystalline  $\text{Bi}_2\text{O}_2\text{Se}$  are of single phase and high purity. The optical properties of the as-prepared  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets are also characterized. As seen from Fig. S1 (Supporting information), the UV–vis–NIR absorption spectrum measured from a diluted  $\text{Bi}_2\text{O}_2\text{Se}$  solution shows strong absorbance in the entire visible light region, and is extended to the region of long wavelength with a decreased absorption. The band gap of the  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets can be estimated according to the equation  $(\alpha h\nu)^{1/2} = A \cdot (h\nu - E_g)$ , where  $\alpha$  is the absorption coefficient,  $h\nu$  is the photon energy,  $A$  is a constant, and  $E_g$  is the band gap energy. Following the absorption spectrum, the dependence of  $(\alpha h\nu)^{1/2}$  on  $h\nu$  is derived and shown in Fig. 2f, which results in a band gap ( $E_g$ ) of 0.9 eV. All these characterization results confirm that the  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets are successfully synthesized.

To explore the effect of solvents, several other solvents are used to substitute ethylene glycol in the synthesis. Figs. 3a–d show the TEM images of the  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets synthesized in different solvents (Samples 1–4). The yielded  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets have a small lateral size of  $\sim 60$  nm (Sample 1, Fig. 3a) when the DI water is used in Solution 1. In comparison, their sizes increase to  $\sim 200$  nm (Sample 2, Fig. 3b) or  $\sim 300$  nm (Sample 3, Fig. 3c) when 0.78 mol/L  $\text{HNO}_3$  or 0.5 mol/L mannitol in DI water is used as the solvent in Solution 1. The use of ethylene glycol in Solution 1 further enlarges the lateral size of  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets to over  $1 \mu\text{m}$  (Sample 4, Fig. 3d). It is likely that  $\text{Bi}^{3+}$  is more stable when  $\text{HNO}_3$ , mannitol, or ethylene glycol are added to reduce the competing hydrolysis of  $\text{Bi}^{3+}$  in water [34]. The more stable solution is beneficial to the growth of nanosheets in the subsequent hydrothermal process. In addition, the high viscosity of ethylene glycol reduces the collision frequency of the ions, leading to less number of seeds in the solution for larger nanosheets [35]. XRD characterizations are used to inspect the phase purity of these four products. As shown in Fig. 3e, the XRD patterns for the samples of different solvent systems (Samples 1, 2 and 4) match well with that of tetragonal  $\text{Bi}_2\text{O}_2\text{Se}$  (JCPDS No. 25-1463), while Sample 3 has some impurities inside. It is observed that the intensities of the diffraction peaks

increase with the increased lateral size of nanosheets, indicating the improved crystallinity. Furthermore, the electrical conductivities are measured for compacted films of  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets from Samples 1–4 (Fig. 3f, see the inset for the schematic diagram of measurement setup). It is shown that the conductivities increase with the increasing size of nanosheets for Samples 1, 2 and 4. Sample 3 has a lower conductivity than Sample 2 despite its sizes, which is owing to the impurities in the products.

The reaction time and temperatures are also tuned to investigate the optimum conditions for the control of the morphologies and the sizes of  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets. Fig. S2a (Supporting information) shows the dependence of the lateral sizes of nanosheets on the different reaction time, while other conditions are kept the same. When the reaction time is 4 h, the lateral sizes of most nanosheets are  $\sim 700$  nm or less (Fig. S3a in Supporting information); when the reaction time is increased to 6 h, the nanosheets can grow to a size of  $\sim 1.4 \mu\text{m}$  (Fig. S3b in Supporting information); when the time is extended to 8 h, the continual growth of nanosheets lead to their aggregation into small bumps or even precipitation, therefore the large-sized nanosheets appear rare in the final dispersion (Fig. S3c in Supporting information). As to the reaction temperature, it has complicated effects on the formation of nanosheets. It is observed that the lateral sizes of the nanosheets synthesized at  $120^\circ\text{C}$  and  $180^\circ\text{C}$  are much smaller than those of the nanosheets synthesized at  $150^\circ\text{C}$  (Figs. S2b and S4 in Supporting information). The formation of  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets is a thermodynamically controlled process. At a relatively low temperature ( $120^\circ\text{C}$ ), the reaction rate is very slow, making the nanosheets unable to grow sufficiently; on the other hand, a high temperature ( $180^\circ\text{C}$ ) makes lateral growth of nanosheets unfavorable due to the increased activity of other crystalline facets. The above discussion shows that an appropriate selection of the reaction time and temperature can promote the growth of nanosheets.

We further study the effect of the amounts of different reactants (NaOH or sodium citrate), or the volume ratio of ethylene glycol and water to understand their effects in the reaction. The increased addition of NaOH from 300 mg to 600 mg leads to less homogenous distribution of sizes, and a drawback of aggregation of nanosheets

(Figs. S2c and S5 in Supporting information). As a conventional mineralizer, an appropriate amount of sodium hydroxide can not only promote the reaction rate, but also has an important effect on the morphology of the products [5,36,37]. But excessive amount of sodium hydroxide accelerates crystal nucleation, resulting in an increase in nucleation sites and a decrease in the size of nanosheets. The coordination agent sodium citrate is also related to the formation of  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets (Fig. S2d in Supporting information). Without any sodium citrate in the reaction,  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets of small sizes are obtained (Fig. S6a in Supporting information); when 2.7 mmol of sodium citrate is added, micron-sized nanosheets are formed (Fig. S6b in Supporting information); when 4.1 mmol of sodium citrate is added, the size of nanosheets decrease (Fig. S6c in Supporting information). In the reaction, sodium citrate chelates with  $\text{Bi}^{3+}$  to form a stable metal complex, and varying amounts of sodium citrate regulate the concentration of free  $\text{Bi}^{3+}$  in the system to balance the coordination and dissolution equilibriums, and further regulate the reaction rate and grain growth [33,38]. The volume ratio of ethylene glycol to water also affects the synthesis of  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets. The optimal ratio is 1:1 with favorable production of high quality and pure phase  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets (Figs. S2e and S7 in Supporting information). If the ratio is larger than 1, products of  $\text{Bi}_2\text{O}_2\text{Se}$  and  $\text{Bi}_2\text{Se}_3$  are harvested (Fig. S7d in Supporting information) with a lateral size decrease (Figs. S7b and c in Supporting information) due to the reduced viscosity of the system. All the above discussions demonstrate that the sizes and morphologies of  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets are affected by a series of reaction conditions.

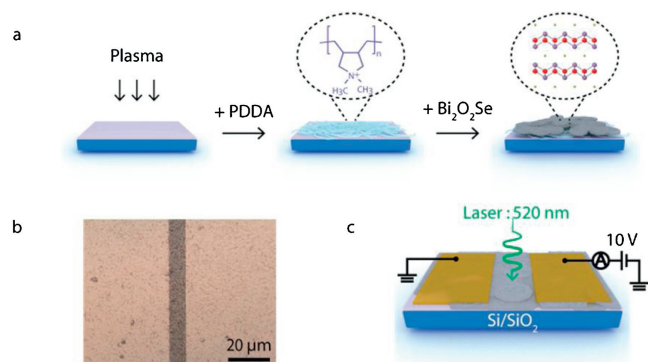
A high-quality thin film of the  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets obtained at optimum reaction conditions is constructed by electrostatic assembly (Fig. 4a). In a typical process, a pretreated substrate is immersed in a positively charged PDDA solution, and then the water dispersion of  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets (for more details, see Experimental Section in Supporting information).  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets are supposed to have negative charges due to the stabilizing citrate ligands, which is verified by the measured surface charge potential of  $-39.3$  mV (Fig. S8 in Supporting information). This large negative surface potential also ensures the great stability of  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets in water. The  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets can be dispersed in the DI water for at least one day (Fig. S8). The optical image in Fig. 4b shows that the  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets are densely assembled on the  $\text{SiO}_2/\text{Si}$  substrate, forming a uniform film without localized aggregation.

This film is then used to fabricate two-terminal photodetectors by typical photolithography processes. The distance between two

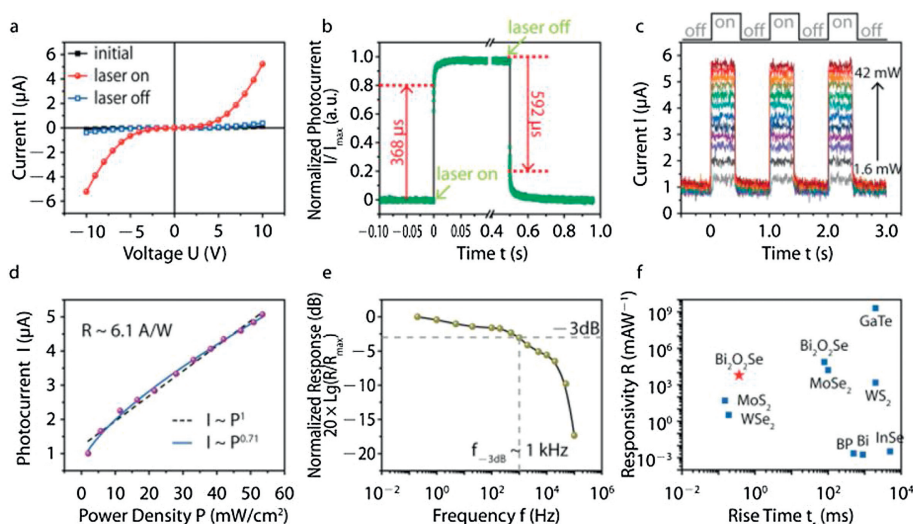
contact electrodes is kept at  $10\ \mu\text{m}$  in the devices. A schematic diagram of the device in the photoelectric experiment is shown in Fig. 4c. The current-voltage ( $I$ - $V$ ) characteristics of the  $\text{Bi}_2\text{O}_2\text{Se}$  films are measured (Fig. 5a). In a dark environment, the original resistance of the sample is about  $72.8\ \text{M}\Omega$  estimated from the current measured at  $10\ \text{V}$  (Fig. 5a). Under the  $520\ \text{nm}$  laser irradiation, the  $I$ - $V$  characteristics have changed significantly, and the current is greater than  $52\ \mu\text{A}$  at  $10\ \text{V}$  (Fig. 5a). The dark currents are recovered after the laser is off (Fig. 5a). The photocurrent  $\Delta I$  is calculated as the variation between the current with and without laser irradiation, *i.e.*,  $\Delta I = I_{\text{ph}} - I_{\text{dark}}$ . A series of photoelectric measurements are then conducted to characterize the responsiveness of  $\text{Bi}_2\text{O}_2\text{Se}$  thin films. In Fig. 5b, the time-dependent characterizations are obtained, shown in the form of normalized photocurrent ( $\Delta I/\Delta I_{\text{max}}$ ) versus time. When the laser is turned on at  $t=0$ , the photocurrent rises sharply from the background current ( $\Delta I/\Delta I_{\text{max}}=0$ ) and rises to the maximum current ( $\Delta I/\Delta I_{\text{max}}=1$ ) within  $0.5\ \text{s}$ . The time of the rising edge is approximately  $368\ \mu\text{s}$  (rise time  $t_r$ , 0%–80%). At the same time, the photocurrent decreases when the laser is turned off. It takes about  $592\ \mu\text{s}$  during the decay edge (decay time  $t_d$ , 100% to 20%), which is slightly longer than the rising edge. Both  $t_r$  and  $t_d$  are very short with a magnitude of  $10^2\ \mu\text{s}$ , therefore the photoelectric curves in Fig. 5b are close to a perfect rectangle, exhibiting good photoelectric performances. These response properties, especially the fast response speed, can be attributed to the large sizes of nanosheets and small number of electronic traps in the  $\text{Bi}_2\text{O}_2\text{Se}$  films.

Furthermore, the multi-periodic dynamic response performance of the  $\text{Bi}_2\text{O}_2\text{Se}$  film is studied, and it shows a high and fast photoelectric responsiveness. As shown in Fig. 5c, the power received by the sample increased from  $1.6\ \text{mW}$  to  $42\ \text{mW}$  under the  $1\text{-Hz}$  modulated laser irradiation. The recorded current gives a response to the modulated laser (Fig. 5c, bottom panel), which corresponds to the applied square wave signals (Fig. 5c, top panel). According to the results in Fig. 5c, the parameter of responsivity  $R$  can be calculated by dividing the variation in current (photocurrent,  $\Delta I$ ) by the total power received on the effective area  $R = \Delta I/(A \times P)$ , where the power density  $P$  is the laser power per unit area, the unit is  $\text{mW}/\text{cm}^2$  (laser power diameter is about  $1\ \text{mm}$ ), and the effective area  $A$  is  $10\ \mu\text{m} \times 120\ \mu\text{m}$ . Therefore, the power density dependence of the photocurrent is shown in Fig. 5d. According to the linear fitting result, the calculated responsivity  $R$  is  $6.1\ \text{A}/\text{W}$  (dashed black line, Fig. 5d). A detailed study of the power law of  $\text{Bi}_2\text{O}_2\text{Se}$  thin film shows that the photocurrent follows a sub-linear relationship with light density,  $\Delta I \sim P^\alpha$ , where  $\alpha = 0.71$  (Fig. 5d). This phenomenon mainly originates from the saturation absorption of  $\text{Bi}_2\text{O}_2\text{Se}$  samples, and has been observed in photodetectors using several other low-dimensional semiconductors [27]. On the other hand, the slight nonlinearity in the power law is caused by carrier recombination *via* the defects and charge impurities within or between the nanosheets. Although the solution-processed  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets have more defects and imperfections than those materials synthesized by CVD or by mechanical exfoliation, they are still potential candidates for high performance photodetectors.

Our  $\text{Bi}_2\text{O}_2\text{Se}$  thin film photodetectors also have a high-frequency response performance (Fig. 5e and Fig. S9 in Supporting information). Fig. S9 shows the measured voltage response from the current amplifier when the photodetectors are under the different laser modulation frequencies from  $0.2\ \text{Hz}$  to  $100\ \text{kHz}$ . When the modulation frequency is low ( $0.2\ \text{Hz} - 20\ \text{Hz}$ ), the response curve is displayed as a rectangle (Figs. S9a–d). Similar to Fig. 5b, the rising edge is always faster than the decay edge, and rise/decay time are around hundreds of microseconds. As the modulation frequency increases, slight distortions tend to appear



**Fig. 4.** Thin-film assembly and photodetectors of  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets. (a) Schematic diagram of the electrostatic assembly process for  $\text{Bi}_2\text{O}_2\text{Se}$  thin films. (b) Optical image of the prepared  $\text{Bi}_2\text{O}_2\text{Se}$  thin film after the deposition of metal electrodes. (c) Schematic for  $\text{Bi}_2\text{O}_2\text{Se}$  thin-film photodetectors under a  $520\text{-nm}$  laser illumination and the electrical connections for measurements.



**Fig. 5.** Photoelectric performances of the  $\text{Bi}_2\text{O}_2\text{Se}$  photodetectors. (a)  $I$ - $V$  characteristics of the  $\text{Bi}_2\text{O}_2\text{Se}$  films with (red balls) and without (blue boxes) a laser illumination, in comparison to that of the initial state (black squares). (b) The time-resolved photoelectric response curve under a laser illumination for 0.5 s and a bias of 10 V. (c) Multi-periodic response curves with a variation of laser power illumination: 1.6, 4.5, 9, 13, 17, 22, 26, 30, 33, 37, 40 and 42 mW. Top: A series of square waves at 1 Hz of the signal generator. (d) Power density ( $P$ ) dependence of the photocurrent ( $\Delta I$ ) in  $\text{Bi}_2\text{O}_2\text{Se}$  samples. The experimental points are fitted following a linear and a nonlinear power law  $\Delta I \sim P^\alpha$ , which are  $\alpha = 1$  (dashed black line) and  $\alpha = 0.71$  (solid blue line), respectively. (e) Correlation of the normalized response and the frequency, showing a  $-3$  dB bandwidth at  $\sim 1$  kHz. (f) Performance comparison of thin-film photodetectors based on different 2D materials synthesized by solution methods [39].

on the curve, but the response amplitude value remains at a similar level to that at relatively low frequencies until it reaches  $10^2$  Hz (Figs. S9e–g). When the frequency is higher than a kilohertz ( $10^3$  Hz), the response curve changes from a rectangle to a distorted rectangle (Figs. S9h–j). When the frequency is as high as 10 kHz or more, the response curve looks like a triangle (Figs. S9k–m). Finally, the response behavior can still be distinguished at 100 kHz (Fig. S9n). Based on these results, we plot the correlation between the normalized responses and the modulation frequency  $f$ , as shown in Fig. 5e. The normalized response is given by  $20 \times \lg(R/R_{\max})$  in dB, where  $R_{\max}$  is regarded as the responsivity obtained at  $f = 0.2$  Hz. The typical parameter is the  $-3$  dB bandwidth at  $f = 1$  kHz, at which the response drops to 70.8 % of the maximum value. Then, the time constant  $\tau$  can be estimated from the  $-3$  dB frequency using the general relationship:  $\tau \sim 0.35 / f_{-3\text{dB}}$ , thus  $\tau \sim 350 \mu\text{s}$ . It is very close to the time constants of  $368 \mu\text{s}$  ( $t_r$ ) and  $592 \mu\text{s}$  ( $t_d$ ) obtained from the direct measurement results in Fig. 5b. Compared with other 2D materials prepared by solution methods (Fig. 5f), our  $\text{Bi}_2\text{O}_2\text{Se}$  photodetectors have higher and faster response. This observation may be due to the large sizes of nanosheets in the films, which allow the photogenerated carriers to move easily. Therefore, we provide a large-scale and low-cost way to develop high-performance photodetectors in large-scale but low-cost production based on the assembled films of colloidal  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets.

In conclusion, we have successfully synthesized the micron-scale and ultrathin  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets through a two-step colloidal synthesis route. A variety of experimental variables have been studied to reveal the roles of reagents in the reaction and related mechanisms in the formation of  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets. The precise control of  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets with a lateral size greater than  $1 \mu\text{m}$  and a thickness of 10 nm has been achieved in optimum conditions. Electrostatic assembly is used to fabricate the high-quality thin films of  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets, which are then applied for photodetectors. These devices have high photoresponsivity and short response time, and respond well to modulations up to 100 kHz. Such performance is superior to those of other devices prepared from solution-processed 2D materials. Our work

provides a promising approach to the synthesis and assembly of  $\text{Bi}_2\text{O}_2\text{Se}$  nanosheets for high-performance photodetectors.

### Declaration of competing interest

The authors report no declarations of interest.

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### Appendix A. Supplementary data

Supplementary material related to this article can be found, in the online version, at doi:<https://doi.org/10.1016/j.ccl.2021.03.039>.

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