



Communication

Au nanowires with high aspect ratio and atomic shell of Pt-Ru alloy for enhanced methanol oxidation reaction

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ARTICLE INFO

Article history:

Received 8 October 2020

Received in revised form 16 November 2020

Accepted 27 November 2020

Available online 4 December 2020

Keywords:

Au nanowires

Pt/Ru submonolayers

Underpotentially deposited

Core-shell nanostructure

Methanol oxidation reaction

ABSTRACT

The methanol oxidation reaction (MOR) is the limiting half-reaction in direct methanol fuel cell (DMFC). Although Pt is the most active single-metal electrocatalyst for MOR, it is hampered by high cost and CO poisoning. Constructing a Pt or Ru monolayer on a second metal substrate by means of galvanic replacement of underpotentially deposited (UPD) Cu monolayer has been shown as an efficient catalyst design strategy for the electrocatalysis of MOR because of the presumed 100% utilization of atoms and resistance to CO poisoning. Herein, we prepared one-dimensional surface-alloyed electrocatalyst from predominantly (111) faceted Au nanowires with high aspect ratio as the substrate of under-potential deposition. The electrocatalyst comprises a core of the Au nanowire and a shell of catalytically active Pt coated by Ru. Coverage-dependent electro-catalytic activity and stability is demonstrated on the Pt/Ru submonolayers on Au wires for MOR. Among all these catalysts, Au@Pt_{ML}@Ru_{ML} exhibits the best electrocatalytic activity and poisoning tolerance to CO. This presents a viable method for the rational catalyst design for achieving high noble-metal utilization efficiency and high catalytic performance.

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As the consumption of traditional fossil fuels increases year by year, and environmental problems becomes increasingly serious, it is a challenging task to find new, clean, sustainable, and renewable energy sources to meet the current demands. Due to its high energy density, widely available and feasible transport of the liquid fuel, direct methanol fuel cell (DMFC) stands out from many fuel cells and become one of the most promising fuel cells [1–6]. Meanwhile, the methanol oxidation reaction (MOR) is a basic anode reaction of DMFCs and the preparation of efficient MOR electrocatalyst remains as an urgent problem in this field [7–12].

Pt nanomaterials have received considerable attention because of their superior catalytic nature in MOR [7,13–17]. Increasing efforts have been devoted to improving the catalytic activity by atom-precise controlled synthesis of Pt catalysts with tailored morphologies [14,17–20]. However, their application is still limited

because of the low abundance and high prices of Pt. The design of core-shell nanostructure catalysts with a shell of Pt or Pt alloy is an effective way to reduce Pt content and retain its high activity [21–26]. One facile approach of constructing a Pt monolayer (Pt_{ML}) on a second metal substrate (such as Pd, Au and Ir) is *via* under-potential deposition (UPD) of Cu monolayer (Cu_{ML}) followed by galvanic replacement with Pt [27–33]. By controlling the amount of charge and deposition voltage in UPD, the coverage of Pt or Ru can be controlled in sub-monolayer (Pt_{s-ML}/Ru_{x-ML}) precision. Meanwhile, the activity of Pt_{ML} electrocatalysts is also dependent on the substrate metal [28,34,35]. The catalytic activity of the surface sites can be modified by manipulation of the local chemical environment through the ligand effect and strain effect [36,37]. For instance, the activity of Pt can be enhanced by tensile strain as in Pt_{ML} on Au (111) and reduced by compressive strain as in Pt_{ML} on Pd (111) [31]. The CO poisoning of catalyst surface is another issue in MOR as Pt is easily poisoned by the reaction intermediate of adsorbed CO (CO_{ads}) from dissociation of methanol molecules to form Pt-CO_{ads} during the electrocatalytic reaction. CO can only be oxidized to CO₂ at a larger overpotential [36]. According to the research on platinum-based catalysts in methanol oxidation electrocatalysis, Pt-Ru bimetallic nanomaterials have been widely recognized as the best CO-tolerant electrocatalysts [38–43].

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According to Watanabe-Motoo bifunctional mechanism, adding an oxophilic metal such as Ru to Pt provides adsorbed hydroxyl groups (OH_{ads}) at a lower potential, which acts as an oxidant to oxidize CO_{ads} on Pt surface to CO_2 , thereby improving the activity of methanol oxidation [36]. A synthetic protocol combining facile preparation of substrate nanostructure with high surface area, and utilization of the bifunctional effect is still missing [44,45].

In this work, we report the $\text{Au@Pt}_{\text{s-ML}}\text{@Ru}_{\text{x-ML}}$ core-shell nanowires (NWs) with ultralow content of Pt and Ru as a superior MOR electrocatalyst. Au NWs with a diameter of 6.8 nm and Boerdijk-Coxeter helix structure is used as the substrate, which features a low-temperature fast preparation, high surface ratio and well-defined (111) facets [46–48]. Au NWs are selected as the substrate because of the high ratio of surface atoms and one-dimensional electron pathway [44,46,47]. The surface coating is realized by successive UPD of Cu on the Au NWs with high aspect ratio followed by galvanic replacement by Pt and Ru while different coverage of Pt and Ru can be controlled by the amount of deposited Cu. The introduction of Ru to surface Pt further boosts the performance of the bimetallic core-shell structure as compared to the pure Pt monolayer on Au NWs. A volcano plot was derived correlating the Ru coverage and the electrocatalytic performance in MOR. Long-term stability of the most optimum catalyst of $\text{Au@Pt}_{\text{ML}}\text{@Ru}_{\text{ML}}$ is also demonstrated. Our results provide a fast but effective method to maximize the precious metals in atomic precision and provide a novel and reasonable way to design MOR electrocatalysts with enhanced activity and durability.

It is acknowledged that one-dimensional Au NWs with high aspect ratio can provide the large surface area, low Ostwald ripening and fast electron communication [42,44]. In this work, the Au NWs were synthesized in a quick protocol and chosen as substrate to construct Au@Pt core-shell NWs. We demonstrate a rapid and feasible fabrication of Au NWs with high aspect ratio (~ 6.8 nm) and yield (approximately 100%). Fig. 1 shows the characterization of the Au NWs obtained after the reduction of HAuCl_4 for 1 min at 60 °C. From the transmission electron microscopy (TEM) image in Fig. 1a, the obtained product is uniform NWs with high aspect ratio, i.e. length of up to several hundreds of nanometers and average diameter of 6.8 nm. The observed dark contrast in Fig. 1b may be caused by the stronger diffraction of the constituent tetrahedral which are oriented close to specific low-index directions. Like the previous report, the Au NWs adopt a Boerdijk-Coxeter (BC) type helix structure (inset of

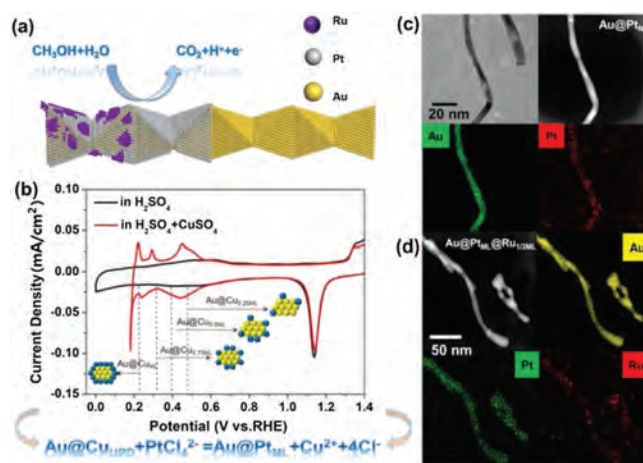


Fig. 2. The model and the characterization of $\text{Au@Pt}_{\text{s-ML}}$ and $\text{Au@Pt}_{\text{ML}}\text{@Ru}_{\text{x-ML}}$. (a) A model illustration of Pt and Ru deposition on the Boerdijk-Coxeter type helical Au nanowire. (b) Cu UPD on the AuNW/CNT in N_2 -saturated $\text{H}_2\text{SO}_4 + 5$ mmol/L CuSO_4 solution, the inset four arrows illustrate the preparation of $\text{Au@Cu}_{\text{s-ML}}$ with different Cu coverages. Scan rate: 5 mV/s. (c) STEM-HAADF image and EDX elemental mapping of Au@Pt_{ML} , suggesting a near monatomic-thick monolayer structure of the Pt_{ML} . (d) HAADF image and EDX elemental mapping of $\text{Au@Pt}_{\text{ML}}\text{@Ru}_{1/3\text{ML}}$.

Fig. 1b) and predominantly expose {111} facets [41]. The HRTEM image (Figs. 1d–f) clearly reveals the dominant {111} facets on the surface of the Au nanowires which are further decorated with rich steps. The measured lattice spacings of ~ 0.232 nm can be assigned to the {111} planes of Au with face-centered cubic (fcc) structure. Furthermore, the varied {111} orientations indicate a possible Boerdijk-Coxeter (BC) type helix structure (Fig. 1b) composed of stacked tetrahedra with twisted orientations. Multiple domains can be identified from the NWs with different orientations, which contributes to the blurred lattice image in the HRTEM image. The X-ray diffraction (XRD) pattern (Fig. 1c) of the NWs exhibits a set of diffraction peaks corresponding to fcc Au.

To obtain $\text{Au@Pt}_{\text{s-ML}}$ with different Pt coverage, Cu was deposited on the Au NWs with controlled coverage via UPD. The UPD deposition of monatomic layer of Cu has been well-developed on single crystalline Au(111) surfaces, but has been rarely observed experimentally on the surface of Au NWs. Fig. 2a shows the cyclic

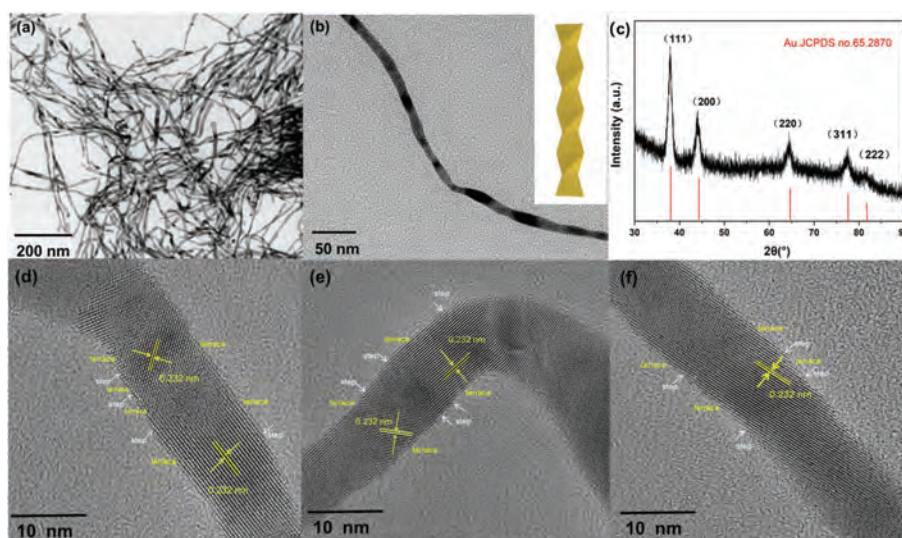
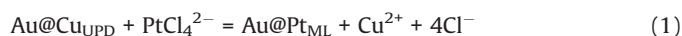


Fig. 1. The characterization of the Au wires with high aspect ratio. (a) Low- and (b) high magnification TEM images of the Au NWs. The inset of (b) shows a model illustration of the Boerdijk-Coxeter type helical nanowire. (c) XRD pattern of the Au NWs. (d) HRTEM images of an individual Au nanowire.

voltammogram (CV) of the Au NWs in N₂-saturated 0.5 mol/L H₂SO₄ and 0.5 mol/L H₂SO₄ containing 5 mmol/L CuSO₄ electrolyte. The reduction peak of Au oxide is constantly located at 1.13 V vs. RHE in both cases. Two additional pairs of redox peaks can be identified from the red curve (between 0.2 V and 0.6 V) before the bulk deposition of Cu, which are attributed to the Cu UPD features at Au (111) surfaces. They were ascribed to two different deposition structures including coadsorption of Cu and sulfate anion followed by a complete Cu adlayer deposition. The electrochemical surface area (ECSA) of the as-synthesized Au NWs catalyst was calculated from the charge associated with the Cu UPD peaks. Assuming a charge constant of 420 μC/cm², the ECSA of the Au NWs was estimated to be around 0.24 cm². NWs of Au@Cu_{s-ML} with different linear scanning voltammetry from 0.9 V to different termination potentials (Fig. 2b). The TEM image of the Au@Cu_{ML} in Fig. S1a (Supporting information) shows similar smooth surface and segmented contrast as the initial Au NWs (Fig. 1a), indicating an ideal Cu_{ML} structure without obvious island formation.

The NWs of Au@Pt_{s-ML} were derived by subjecting the above-mentioned NWs of Au@Cu_{s-ML} to galvanic replacement with K₂PtCl₄ in N₂-saturated 0.5 mol/L H₂SO₄. In this process, the Pt shell is deposited on the surface of Au NWs through the displacement with the Cu shell, which can be shown in the following formula:



The Au@Pt_{s-ML} catalyst was characterized by TEM, HAADF-STEM and energy-dispersive X-ray spectroscopy (EDX) elemental mapping, which reveals that monolayer Pt shell was uniformly coated on the surface of the Au NWs (Fig. 2c). On the basis of the Au@Pt core-shell nanostructures, different layers of Ru were further deposited on these core-shell NWs via the UPD of Cu and galvanic replacement with RuCl₃. As shown in Fig. S2 (Supporting information), the cyclic voltammogram (CV) of the Au@Pt_{ML} NWs is obtained in N₂-saturated 0.5 mol/L H₂SO₄ containing 5 mmol/L CuSO₄ electrolyte. NWs of Au@Pt_{ML}@Ru_{x-ML} with different Ru coverage ($x = 1/3, 2/3, 1$ and $4/3$) can be obtained by negative linear scanning voltammetry from 0.9 V to different termination potential in the above solution followed by adding 25 mL of RuCl₃ (5 mmol/L) into the solution. The distribution of Ru, Pt and Au in the NWs are provided by the STEM image and EDX elemental mapping (Fig. 2d and Fig. S3 in Supporting information) of Au@Pt_{ML}@Ru_{1/3ML} NWs and Au@Pt_{ML}@Ru_{ML} NWs (Fig. S4 in Supporting information). By comparing the EDX data, the amount of Pt is almost the same and the amount of Ru of Au@Pt_{ML}@Ru_{ML} is three times as much as Au@Pt_{ML}@Ru_{1/3ML}, which is in good agreement with the electrochemistry data. While Pt-Ru alloyed catalysts usually exhibit high activity and enhanced durability toward MOR, systematic study of the structure-dependent electrocatalytic properties of Au@Pt_{s-ML} and Au@Pt_{ML}@Ru_{x-ML} core-shell NWs is conducted to optimize the catalytic performance. Figs. 3a and c show MOR activity associated with different Au@Pt_{s-ML} and Au@Pt_{ML}@Ru_{x-ML} catalysts measured in 0.1 mol/L HClO₄ aqueous solution with 0.5 mol/L CH₃OH at a sweep rate of 50 mV/s, and the MOR oxidative current density at 0.8 V vs. RHE are compared in Figs. 3b and d. As the nominal Pt coverage increases from 0.25 ML to 1 ML (as indicated in Fig. 3b), the MOR activity increases accordingly.

This is fully consistent with the trend in the ECSA_{Pt} shown in Fig. S5 (Supporting information), indicating that the MOR performance directly correlates with ECSA_{Pt}. Similar trend can be derived by comparing the magnitude of the Pt-oxide reduction peaks. For these metal overlayers, the activity improvement originates from a combination of Au-Pt ligand effects and local

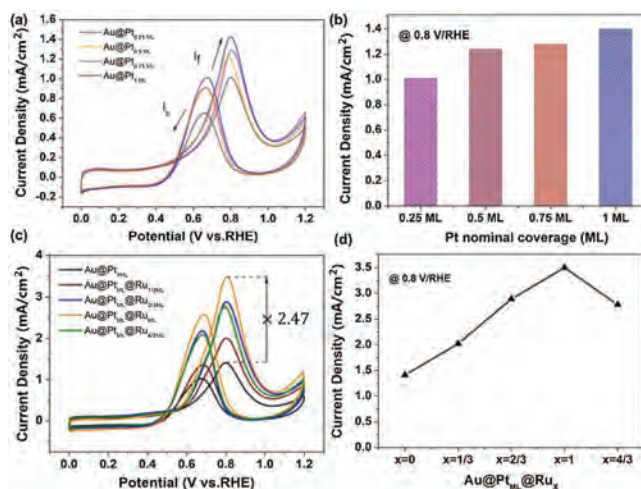
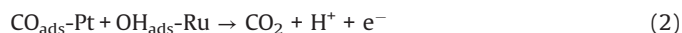


Fig. 3. MOR performance of Au@Pt_{s-ML} and Au@Pt_{ML}@Ru_x. (a, c) The CV of Au@Pt_{s-ML} and Au@Pt_{ML}@Ru_{x-ML} in 0.1 mol/L HClO₄ + 0.5 mol/L CH₃OH (at 50 mV/s scan rate). (b, d) Comparison of oxidative MOR current density at 0.8 V vs. RHE.

strain effects manipulated by the underlying Au(111) lattice. The Au NWs have a Boerdijk-Coxeter helix-type structure dominated by (111) facets. As studied by Adzic *et al.*, Pt_{ML}/Au(111) with tensile strain exhibits over seven-fold enhancement in peak current density with respect to Pt(111) (the most active low-index plane of Pt) [31]. In addition, the electronic interaction between Pt_{ML} and the Au substrate results in a weakened binding strength of CO_{ads}, and efficiently catalyze the CO_{ads} oxidation, which is the rate-limiting step in the indirect mechanism of the MOR.

It is widely acknowledged that the MOR could go through a dual-path mechanism: direct pathway (reactive intermediates) and indirect pathway (poisoning intermediates). This is consistent with previous reports that the electrochemical oxidation of methanol on Pt-based surfaces preferentially through the indirect pathway, where methanol molecules sequentially undergoes dissociative adsorption, dehydrogenation to form CO_{ads} and oxidation of the CO_{ads}, the last step of which is considered to be the rate-limiting step. And the poisoning intermediates are determined mainly as adsorbed CO_{ads}, which would be hardly stripped out until the OH_{ads} is generated on the Pt surfaces under high electrode potential. Fortunately, the additional Ru atoms can promote water oxidation to increase the availability of OH_{ads} at lower overpotential and consequently enhance the activity via the bifunctional Langmuir-Hinshelwood mechanism:



As shown in Fig. 3c, Au@Pt_{ML}@Ru_{ML} exhibits a negative shift in onset potential and 2.47 times enhancement in peak current density at 0.8 V/RHE with respect to Au@Pt_{ML}, establishing it as the best-performing catalyst among all the samples. Previous studies by Mavrikakis *et al.* demonstrated the free energy change of CO adsorption as one of the key descriptors, which determines the onset potential for MOR. So the negative shift of the onset potential toward MOR on Au@Pt_{ML}@Ru_{ML} NWs agrees well with the finding and the higher I_f/I_b (forward and backward current density respectively) ratio implies that the methanol could be effectively oxidized during the forward potential scan, producing less poisoning species, thereby possessing higher tolerance of CO poisoning. We also found that the activity based on the weight of Pt and all metals (including Au, Pt, Ru) shows similar trend for different alloy catalysts and the performance of the best one Au@Pt_{ML}@Ru_{ML} is superior to that of many similar alloy catalysts in recent studies (Fig. S6 and Table S1 in Supporting information). To

further investigate the structure–performance relationship in Au–Pt–Ru ternary system, the current density of nanowires of bare Au, Au@Ru_{ML} and Au@Ru_{ML}@Pt_{ML} is evaluated under the same experimental condition, as shown in Fig S7 (Supporting information). The nanowires of bare Au and Au@Ru_{ML} have negligible activity in catalyzing MOR while Au@Ru_{ML}@Pt_{ML} exhibits much higher activity. When a monolayer of Ru atoms is deposited on Au NWs firstly, they tend to form islands on Au surface, and the following Pt deposit on the uncovered Au surface, resulting in a similar structure of Ru–Pt interface on Au. However, it is still lower than the performance of Au@Pt_{ML}@Ru_{ML}. The inferior performance may come from the direct landing of Ru on Au and less available Au surface for Pt deposition. Hence the atom-precision fine-tune of Au@Pt_{ML}@Ru_{ML} provides the most favorable catalyst for MOR.

Similar to the trend in Au@Pt_{s-ML}, the current density for MOR were found to depend on the surface coverage of Ru atoms, exhibiting an increasing catalytic activity with increasing Ru coverage, until reaching a maximum at nominal surface Ru coverage of 1 ML. From the CV of Au@Pt_{ML}@Ru_{ML} in N₂-saturated 0.1 mol/L HClO₄ solution, the peak of Au oxidation at 1.17 V vs. RHE nearly disappears as compared to the inset figure of Au NWs in 0.5 mol/L H₂SO₄, suggesting the surface of the electrocatalyst was covered by Pt–Ru atoms (Fig. 4a). However, the activity began to subsequently decrease when more than 1 ML of Ru is deposited on the surface (Fig. 3d). The volcano trend in the MOR activity suggests a remarkable impact of the Ru amount on the catalytic activity of core–shell nanostructures. With the increase of Ru deposition, the ligand effect on the underlying Pt layer will be enhanced, which contributes to lower the d-band center of Pt and weaken the CO adsorption on Pt. However, less Pt sites will be exposed in this case. The best performance come with a balance between enhanced intrinsic activity and enough active sites and hence with intermediate coverage of Ru on the surface of the Au@Pt_{ML} nanowires. Actually, the peak position of the volcano plot indicates an incomplete coverage of the substrate with nominal one monolayer of Ru.

The effect induced by alloying of Ru and Pt is further evaluated by CO stripping voltammetry. As shown in Fig. 4b, the CO stripping potential for Au@Pt_{ML}@Ru_{ML} is slightly lower than that of Au@Pt_{ML},

in agreement with previously established results (Fig. 3c), consolidating that the d-band center of the Pt layer is reduced by the ligand effect imposed by Ru. As shown in Fig. 3c, the ratio of I_f to I_b for Au@Pt@Ru_{ML} is apparently larger than that for Au@Pt_{ML}, indicating less poisoning effect for Au@Pt_{ML}@Ru_{ML} than for Au@Pt_{ML}, in agreement with the conclusion derived from the CO stripping (Fig. 4b). However, the down-shift of the d-band center also reduces the adsorption energy and surface coverage of CO species on the catalyst and lowers the catalytic activity. Only when the two effects strike a balance, the sample of Au@Pt_{ML}@Ru_{ML} achieves the optimum MOR performances and a volcano-shape dependence of the catalytic activity on the coverage is observed.

The stability or durability of the catalyst is the most critical requirement of catalysts for practical application of fuel cells. The above-mentioned catalysts were further evaluated through an accelerated durability test (Fig. 4c). The peak current density of methanol oxidation obtained from forward CV sweep decreases gradually with the increase of the cycle number. After 800 cycles, the Au@Pt_{ML}@Ru_{ML} NWs retains 62.4% of the initial catalytic activity. This phenomenon may result from of methanol consumption during the successive scans and accumulation poisonous species on the surface of the electrocatalyst. Actually the consumption of methanol in the whole process is very low. So it is suggested that the main reason for the decrease of current density in the 800 cycles may be the accumulation poisonous species on the surface of the electrocatalyst. Hence, the catalyst was evaluated again after 800 cycles in freshly prepared electrolyte (0.5 mol/L CH₃OH+0.1 mol/L HClO₄) and the peak current density was 3.24 mA/cm², equivalent to 92.4% of the initial value (Figs. 4c and d, Fig. S8 in Supporting information). The fresh electrolyte solution provides the diffusion conditions for the poisonous species gathered on the electrocatalyst surface, and leads to the recovery of the MOR performance. The excellent long-term cycle stability of the multilayer nanostructure further reveals that Au@Pt_{ML}@Ru_{ML} catalyst may be a good alternative catalyst in DMFC.

In summary, we successfully synthesized nanowires of Au@Pt_{s-ML} and Au@Pt_{ML}@Ru_{x-ML} with ultralow loading of noble metals and high MOR performance *via* UPD and galvanic replacement. The enhanced MOR activities of these multilayer structures can be attributed to the Au–NW substrate as an electron pathway for Pt monolayer and ligand effect from Ru, which decreases the adsorption energy by lowering the d-band center of Pt. The Ru sites also promote CO oxidation on Pt *via* the enhanced adsorption of OH groups, as supported by the CO stripping test. We present a correlation between the electrocatalytic activity of the NWs and the surface coverage of noble metals. The optimum Ru coverage in Au@Pt_{ML}@Ru_{ML} comes at the balance between exposing Pt active sites and imposing ligand effect and bifunctional mechanism by Ru coating. This avenue may shed light on the rational design of cost-efficient electrocatalysts for the portable DMFC in the future.

Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Acknowledgments

This work was financially supported by the Natural Science Foundation of Tianjin, China (No. 18JCYBJC20600), the National Natural Science Foundation of China (Nos. 62074123, 61701543) and Institute of Energy, Hefei Comprehensive National Science Center (No. 19KZS207).

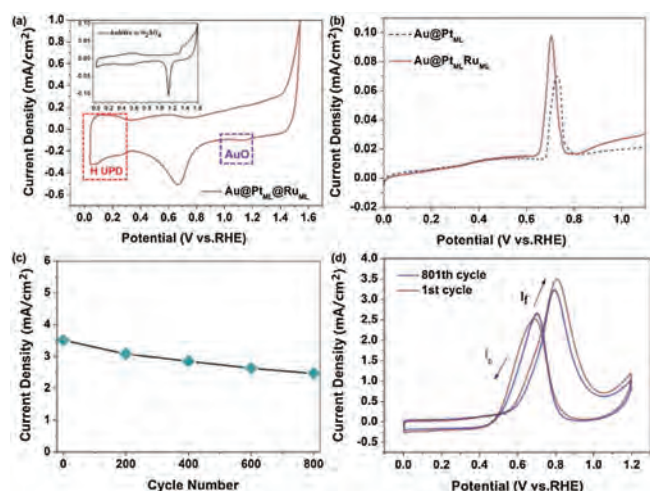


Fig. 4. The electrochemical characterization and stability test. (a) CVs of Au@Pt_{ML}@Ru_{ML} in N₂-saturated 0.1 mol/L HClO₄ solution at 50 mV/s, the inset figure depicts CV of Au NWs in 0.5 mol/L H₂SO₄. (b) The CO stripping current measured *via* potential sweep at 5 mV/s in 0.1 mol/L HClO₄ (adsorptions at 50 mV). Only the positive-going sweeps are shown for clarity. (c) The relationship between the peak current density and the cycle number. (d) CVs of Au@Pt_{ML}@Ru_{ML} before and after 800 potential cycles. The 801th cycle in freshly prepared 0.5 mol/L CH₃OH + 0.1 mol/L HClO₄ solution, scan rate: 50 mV/s.

Appendix A. Supplementary data

Supplementary material related to this article can be found, in the online version, at doi:<https://doi.org/10.1016/j.ccl.2020.11.071>.

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