



## Communication

# A unique 3D microporous MOF constructed by cross-linking 1D coordination polymer chains for effectively selective separation of CO<sub>2</sub>/CH<sub>4</sub> and C<sub>2</sub>H<sub>2</sub>/CH<sub>4</sub>

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## ABSTRACT

Selective separation of CO<sub>2</sub>/CH<sub>4</sub> and C<sub>2</sub>H<sub>2</sub>/CH<sub>4</sub> are promising for their high-purity industrial demand and scientific research on account of the similar molecular radius and physical properties. In this work, a unique 3D microporous MOF material [Cu(SiF<sub>6</sub>)(sdi)<sub>2</sub>]-solvents (**1**, sdi = 1,1'-sulfonyldiimidazole) was successfully constructed by cross-linking 1D coordination polymer chains. The dense functional active sites on the inner walls of the channel of **1a** can provide strong binding affinities to CO<sub>2</sub>, C<sub>2</sub>H<sub>2</sub>, and thus effectively improve the gas separation performance of CO<sub>2</sub>/CH<sub>4</sub> and C<sub>2</sub>H<sub>2</sub>/CH<sub>4</sub>.

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Selective separation of carbon dioxide (CO<sub>2</sub>) and acetylene (C<sub>2</sub>H<sub>2</sub>) from methane (CH<sub>4</sub>) are of significant challenges in natural gas purification and industrial acetylene production [1,2]. However, the conventional porous materials such as zeolites and porous carbons, polymer-based membranes and cryogenic distillation technology cannot meet the requirements for selective separation of the impurities with the similar molecular kinetic diameter (CO<sub>2</sub>: 3.3 Å, C<sub>2</sub>H<sub>2</sub>: 3.3 Å, CH<sub>4</sub>: 3.758 Å), boiling point (CO<sub>2</sub>: 216.55 K, C<sub>2</sub>H<sub>2</sub>: 188.40 K, CH<sub>4</sub>: 111.66 K) and polarizability (CO<sub>2</sub>: 29.11 × 10<sup>25</sup> cm<sup>-3</sup>, C<sub>2</sub>H<sub>2</sub>: 33.3–39.3 × 10<sup>25</sup> cm<sup>-3</sup>, CH<sub>4</sub>: 25.93 × 10<sup>25</sup> cm<sup>-3</sup>) [3]. Moreover, the low efficiency and high energy consumption during the separation process also made them less compatible [4]. Therefore, the development of new materials for CO<sub>2</sub>/CH<sub>4</sub> and C<sub>2</sub>H<sub>2</sub>/CH<sub>4</sub> separation are of great needs.

Metal-organic frameworks (MOFs), as crystalline porous materials, have been widely studied in the areas of gas storage and separation [5–9], luminescent sensing [10–12], heterogeneous catalysis and electrochemistry [13–15]. Benefiting from their unique structures, MOFs are regarded as ideal materials for effectively binding and selective separation of CO<sub>2</sub>/CH<sub>4</sub> and C<sub>2</sub>H<sub>2</sub>/CH<sub>4</sub>. Moreover, the rich active heteroatoms sites and open metal sites (OMSs) can enhance the preferential binding of specific guest molecules. For instance, He *et al.* reported a rare Pb<sub>9</sub>-cluster MOF

based on a flexible cyclotriphosphazene-functionalized hexacarboxylate ligand for gas separation [16]. Owing to the rich active heteroatoms sites and OMSs on the inner pore surfaces of the MOF, the adsorption selectivities of the MOF for the equimolar CO<sub>2</sub>/CH<sub>4</sub> and C<sub>2</sub>H<sub>2</sub>/CH<sub>4</sub> at 298 K and 100 kPa calculated by the predicted ideal adsorbed solution theory (IAST) are 5.8 and 21.0, respectively. However, it should be noted that the rational predesign and construction of MOF materials containing specific microporous and multiple functional active sites (OMS, -NH<sub>2</sub>, -F, -OH and *etc.*) are still challenging [17–19]. On one hand, the various coordination modes of both center metal ions and organic ligands make it complicated to predict the final structures of the self-assembly MOFs; On the other hand, the obtained MOF materials may not be applicable to the gas separation on account of the unsuitable aperture of MOF channels and the lack of functional active sites for binding with gas molecules.

In order to tackle the intractable problem, we promoted a cross-linking strategy for the construction of porous MOF materials with multifunctional active sites for effectively selective separation of gas mixtures. A previously reported coordination polymer (CP) [Cu(sdi)<sub>2</sub>(NO<sub>3</sub>)<sub>2</sub>]-2CO(CH<sub>3</sub>)<sub>2</sub> was a representative example for illustrating the strategy [20]. In the titled CP, the one-dimensional (1D) infinite coordinate chains are held together to form three-dimensional (3D) supramolecular structure *via* hydrogen-bonding and Van der Waals interactions. It is worth noting that these 1D chains possess large amounts of terminal coordinated NO<sub>3</sub><sup>-</sup> moieties with the hexa-coordinated metal Cu(II) ions. Inspired by this, if the initial metal salt was changed from Cu(NO<sub>3</sub>)<sub>2</sub>·3H<sub>2</sub>O to

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$\text{CuSiF}_6 \cdot n\text{H}_2\text{O}$ , then the  $\text{SiF}_6^{2-}$  moieties were expected to instead of the terminal  $\text{NO}_3^-$ , and ultimately these separated 1D chains would be cross-linked by the  $\text{SiF}_6^{2-}$  moieties to build a novel microporous 3D network (Scheme 1). In addition, the  $\text{SiF}_6^{2-}$  moieties can also act as functional active sites to facilitate the selective separation of gas molecules, which is proved by previously reported works [21,22].

With this in mind, we successfully constructed a 3D microporous MOF material  $[\text{Cu}(\text{SiF}_6)(\text{sdi})_2]\text{-solvents}$  (**1**). Compound **1** possessed one-dimensional (1D) narrow channels along the *c* direction, and large amounts of O and F donor sites originated from  $-\text{SO}_2-$  and  $\text{SiF}_6^{2-}$  on the inner walls of the channels. These dense functional active sites can provide strong binding affinities to specific guest molecules, and thus effectively improve the gas separation performance. Gas adsorption experiments proved that **1a** (activated **1**) has high adsorption capacity and selectivity for  $\text{CO}_2/\text{CH}_4$  and  $\text{C}_2\text{H}_2/\text{CH}_4$ . In addition, the MOF exhibits high stability not only in common organic solvents but also in deionized water, which is beneficial to its practical and industrial applications.

Single-crystal X-ray diffraction measurement reveals that compound **1** crystallized in the trigonal system with the  $R\bar{3}c$  space group. The asymmetric unit of **1** contains one crystallographically independent sdi ligand, one half of  $\text{Cu}(\text{II})$  cation and  $\text{SiF}_6^{2-}$  anion. As shown in Fig. 1a, the  $\text{Cu}(\text{II})$  centre is hexacoordinated with four nitrogen atoms originated from four different sdi ligands and two fluorine atoms provided by two  $\text{SiF}_6^{2-}$  anions. The Cu-N and Cu-F bond distances are in the normal range of 1.988(2)–2.4472(19) Å, which are comparable to the other similar MOFs reported in the previously published papers [23–25]. The selected bond length and angles are listed in Table S2 (Supporting information). Each sdi ligand is coordinated with two  $\text{Cu}(\text{II})$  ions and connected with another one sdi ligand to generate a  $[\text{Cu}_2\text{sdi}_2]$  unit, and the adjacent units by sharing  $\text{Cu}(\text{II})$  ions to form an infinite 1D chain (Fig. 1b). Finally, the 1D chains are further cross-linked by  $\text{SiF}_6^{2-}$  groups to form a 3D honeycomb-like structure with 1D narrow channels along the *c* axis (Figs. 1c and d). The pore limiting diameter of **1** is around 3.4 Å viewed from *c* axis and the potentially free volume (without the solvent molecules) estimated by PLATON program is 25.9%. In order to better understand the complicated structure of **1**, ToposPro software is applied to simplify the specific structure [26]. Based on the simplification, compound **1** can be regarded as a novel 4-connected unimodal *nbo* PLATON program is 25.9 of  $\{6^4, 8^2\}$  (Fig. 1e).

The purity of **1** was proved by PXRD. As shown in Fig. S2 (Supporting information), the observed PXRD pattern matched well to the simulated one, indicating the bulk sample of **1** possessed good phase purity. TGA curve reveals that **1** has moderate thermal stability and can keep its structure until 230 °C. In addition, **1** can maintain its pristine crystal structure

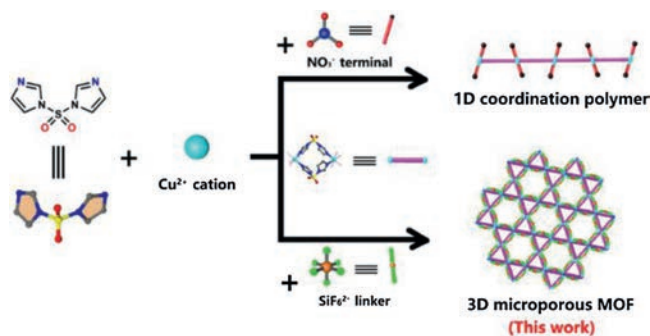
not only in common organic solvents (MeOH, EtOH,  $\text{CH}_3\text{COCH}_3$ , DMF, DMAC, DMSO,  $\text{CH}_3\text{CN}$ ,  $\text{CH}_2\text{Cl}_2$ ,  $\text{CHCl}_3$ ), but also in deionized water for three days, which enhances its potential for practical application (Fig. S3 in Supporting information).

Based on the good stability of **1**,  $\text{N}_2$  adsorption experiment was performed at 77 K to evaluate the porosity of **1a**. As illustrated in Fig. S4 (Supporting information), negligible  $\text{N}_2$  was adsorbed. The sample was collected after the  $\text{N}_2$  sorption measurement for further inspection. PXRD testing confirmed that the framework of **1a** did not collapse (Fig. S2), so the low  $\text{N}_2$  uptake of **1a** can be attributed to the repulsion of the framework to  $\text{N}_2$  and the small aperture size hindered the  $\text{N}_2$  molecules into the channels of **1a** under this circumstance. Subsequently, we changed the probe molecule to  $\text{CO}_2$  because of its small kinetic diameter (3.30 Å) compared to  $\text{N}_2$  (3.64 ~ 3.80 Å). As expected, a high uptake amount of  $\text{CO}_2$  with type I isotherm was observed at 195 K (saturated capacity: 101.2  $\text{cm}^3/\text{g}$ , Fig. S4). The experimental pore volume of **1a** is 0.1512  $\text{cm}^3/\text{g}$ . The Brunauer-Emmett-Teller (BET) surface area and Langmuir surface area of **1a** were calculated to be 358  $\text{m}^2/\text{g}$  and 442  $\text{m}^2/\text{g}$  based on the  $\text{CO}_2$  adsorption experiment, respectively [27,28].

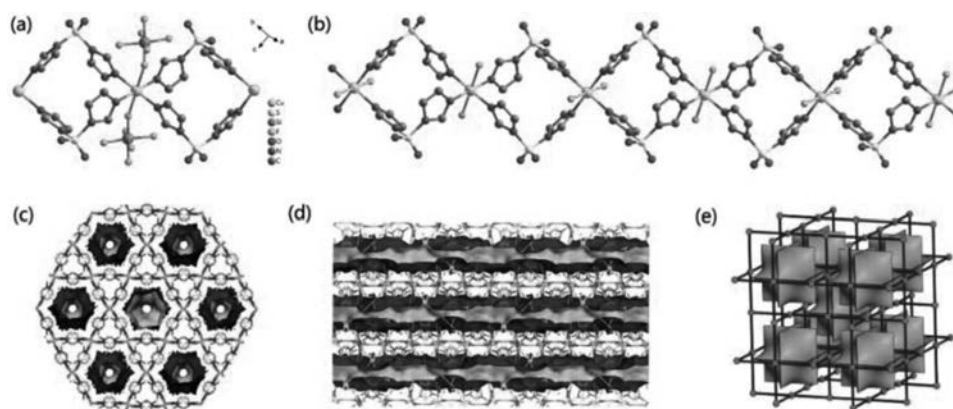
The microporous structure with relatively high BET surface area, multiple functional active sites and high stability prompted us to investigate the gas sorption and separation properties of **1a**.  $\text{CO}_2$ ,  $\text{C}_2\text{H}_2$  and  $\text{CH}_4$  were selected as the targeted molecules on account of their high-purity industrial demand and scientific research. The single-component adsorption experiments of **1a** toward  $\text{CO}_2$ ,  $\text{C}_2\text{H}_2$  and  $\text{CH}_4$  were measured at 273 K and 298 K, respectively. As illustrated in Fig. 2a, **1a** has relatively high  $\text{CO}_2$  and  $\text{C}_2\text{H}_2$  adsorption capacity (29.2  $\text{cm}^3/\text{g}$  for  $\text{CO}_2$  and 34.2  $\text{cm}^3/\text{g}$  for  $\text{C}_2\text{H}_2$ ) at 298 K and 1 atm. When the temperature is reduced to 273 K, the trend is maintained and the corresponding gas uptakes increased to 45.1 and 48.7  $\text{cm}^3/\text{g}$ , respectively (Fig. 2b). Compared to  $\text{CO}_2$  and  $\text{C}_2\text{H}_2$ , only limited amounts of  $\text{CH}_4$  are adsorbed (6.8  $\text{cm}^3/\text{g}$  at 298 K and 11.7  $\text{cm}^3/\text{g}$  at 273 K). The distinctly different gas adsorption behaviors indicated that **1a** can act as a promising material for the selective separation of  $\text{CO}_2/\text{CH}_4$  and  $\text{C}_2\text{H}_2/\text{CH}_4$ .

The adsorption selectivities of **1a** towards binary gas mixtures  $\text{CO}_2/\text{CH}_4$  (50/50, v/v) and  $\text{C}_2\text{H}_2/\text{CH}_4$  (50/50, v/v) were evaluated by the IAST. The Langmuir-Freundlich equation fits extremely well with the single-component isotherms of  $\text{CO}_2$ ,  $\text{C}_2\text{H}_2$  and  $\text{CH}_4$  at 273 and 298 K (Figs. S5–S7 in Supporting information). As shown in Fig. 3, the IAST adsorption selectivities are 10.4 for  $\text{CO}_2/\text{CH}_4$  and 11.2 for  $\text{C}_2\text{H}_2/\text{CH}_4$  at 298 K. As the temperature lowered to 273 K, the selectivities increased to 12.5 and 17.1, respectively. The IAST selectivities of **1a** towards  $\text{CO}_2/\text{CH}_4$  (50/50, v/v) and  $\text{C}_2\text{H}_2/\text{CH}_4$  (50/50, v/v) under 298 K are superior to some previously reported MOFs (Table S3 in Supporting information). These results further proved that **1a** is capable for selective separation of  $\text{CO}_2/\text{CH}_4$  and  $\text{C}_2\text{H}_2/\text{CH}_4$  mixtures.

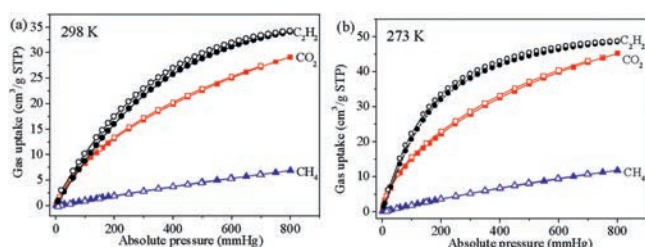
To elucidate the reason that of compound **1a** exhibits comparatively high adsorption amount and selectivity for  $\text{CO}_2$  and  $\text{C}_2\text{H}_2$ , the  $Q_{\text{st}}$  for **1a** toward the different gas molecules were calculated by the Clausius-Clapeyron equation using isotherm fits to the Virial equation based on the single component gas adsorption isotherms measured at 273 K and 298 K (Figs. S8–S10 in Supporting information). The  $Q_{\text{st}}$  values at near zero coverage are found to be 31.9, 26.8 and 16.3 kJ/mol for  $\text{CO}_2$ ,  $\text{C}_2\text{H}_2$  and  $\text{CH}_4$ , respectively (Fig. S11 in Supporting information). Obviously, the  $Q_{\text{st}}$  values of  $\text{CO}_2$  and  $\text{C}_2\text{H}_2$  are much higher than that of  $\text{CH}_4$ , which confirmed the strong affinities between  $\text{CO}_2$ ,  $\text{C}_2\text{H}_2$  molecules and the framework of **1a**, and thus resulting the highly selective separation of  $\text{CO}_2/\text{CH}_4$  and  $\text{C}_2\text{H}_2/\text{CH}_4$ . The  $Q_{\text{st}}$  curves for  $\text{CO}_2$  and  $\text{C}_2\text{H}_2$  showed two distinctly different trends, with the increase of gas adsorption, the  $Q_{\text{st}}$  value for  $\text{CO}_2$  decreased, while that for  $\text{C}_2\text{H}_2$



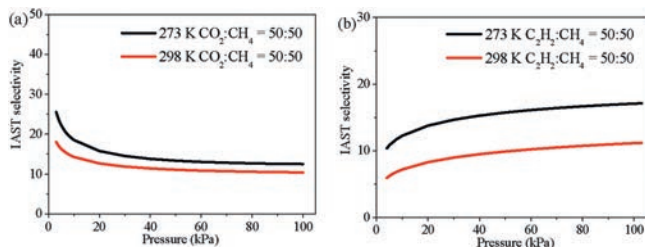
Scheme 1. Schematic for the cross-linking construction of the 3D microporous MOF.



**Fig. 1.** (a) The coordination environment of the centre Cu(II) ion. (b) The infinite 1D Cu(II)-sdi chain (SiF<sub>6</sub><sup>2-</sup> groups are omitted for clarity). (c, d) The 3D structure of **1** along *c* and *b* axes (the isosurfaces represent the voids inside the pores). (e) The tiling image of the simplified framework of **1**. Color codes: Cu (sky blue), S (yellow), Si (orange), F (green), O (red), N (mazarine), C (gray). Hydrogen atoms are omitted for clarity.



**Fig. 2.** Single-component adsorption isotherms of C<sub>2</sub>H<sub>2</sub> (black), CO<sub>2</sub> (red), CH<sub>4</sub> (blue) of **1a** at (a) 298 K and (b) 273 K.



**Fig. 3.** The IAST adsorption selectivities of CO<sub>2</sub>/CH<sub>4</sub> and C<sub>2</sub>H<sub>2</sub>/CH<sub>4</sub> of **1a** at 273 K (black) and 298 K (red).

increased. These phenomena could be ascribed to the continuously variational intermolecular interactions along with the gas adsorption process [29]. Moreover, the  $Q_{st}$  values of both CO<sub>2</sub> and C<sub>2</sub>H<sub>2</sub> are comparable to or even surpassing to some well-known MOFs decorated with functional group sites (-NH<sub>2</sub>, -F, -OH, etc.) and/or OMSs (Table S4 in Supporting information).

Therefore, on the basis of the above analysis, the following aspects can be used as explanation of the preferable adsorption phenomena of **1a**: (a) The small apertures and the repulsion of framework hindered the CH<sub>4</sub> molecules into the pore channels of **1a** [30]; (b) The O and F donor sites originated from the functional active sites (-SO<sub>2</sub><sup>-</sup> and SiF<sub>6</sub><sup>2-</sup>) on the inner walls of the channels can provided strong binding affinities *via* hydrogen-bonding and/or Van der Waals interactions with the CO<sub>2</sub> and C<sub>2</sub>H<sub>2</sub> molecules [31,32]; (c) CO<sub>2</sub> and C<sub>2</sub>H<sub>2</sub> are more easily to be condensed than CH<sub>4</sub> for their relatively higher critical temperature (CO<sub>2</sub>: 304.12 K, C<sub>2</sub>H<sub>2</sub>: 308.3 K, CH<sub>4</sub>: 190.56 K) [33]; (d) The higher polarizability of CO<sub>2</sub> and C<sub>2</sub>H<sub>2</sub> than CH<sub>4</sub> resulted in stronger interactions between the framework of **1a** and CO<sub>2</sub> or C<sub>2</sub>H<sub>2</sub>. All these factors contribute to

the preferable adsorption of **1a** toward CO<sub>2</sub> and C<sub>2</sub>H<sub>2</sub> to CH<sub>4</sub> [34,35].

In summary, a novel 3D MOF material **1** with 1D microporous was rationally designed and constructed by cross-linking 1D coordination polymer chains. The dense functional active sites on the inner walls of the channels can offer strong binding affinities to specific guest molecules, and thus favorable for the gas separation performance. Furthermore, gas adsorption experiments confirm that **1a** have relatively high CO<sub>2</sub> and C<sub>2</sub>H<sub>2</sub> adsorption capacities over CH<sub>4</sub> at 1 bar. The predicted IAST adsorption selectivities for CO<sub>2</sub>/CH<sub>4</sub> and C<sub>2</sub>H<sub>2</sub>/CH<sub>4</sub> are 10.4, 11.2 at 298 K and 12.5, 17.1 at 273 K. This work not only reported a porous MOF for effectively selective separation of CO<sub>2</sub>/CH<sub>4</sub> and C<sub>2</sub>H<sub>2</sub>/CH<sub>4</sub>, but also opened a new perspective for the rational predesign and functionalization of MOFs materials.

### Declaration of competing interest

The authors report no declarations of interest

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### Appendix A. Supplementary data

Supplementary material related to this article can be found, in the online version, at doi:<https://doi.org/10.1016/j.ccl.2020.09.014>.

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