



## Communication

## High performance ethylene sensor based on palladium-loaded tin oxide: Application in fruit quality detection



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## ABSTRACT

Ethylene (C<sub>2</sub>H<sub>4</sub>), as a plant hormone, its emission can be served as an indicator to measure fruit quality. Due to the limited physicochemical reactivity of C<sub>2</sub>H<sub>4</sub>, it is a challenge to develop high performance C<sub>2</sub>H<sub>4</sub> sensors for fruit detection. Herein, this paper presents a resistive-type C<sub>2</sub>H<sub>4</sub> sensor based on Pd-loaded tin oxide (SnO<sub>2</sub>). The C<sub>2</sub>H<sub>4</sub> sensing performance of proposed sensor are tested at optimum operating temperature (250 °C) with ambient relative humidity (51.9% RH). The results show that the response of Pd-loaded SnO<sub>2</sub> sensor (11.1, R<sub>a</sub>/R<sub>g</sub>) is about 3 times higher than that of pristine SnO<sub>2</sub> (3.5) for 100 ppm C<sub>2</sub>H<sub>4</sub>. The response time is also significantly shortened from 7 s to 1 s compared with pristine SnO<sub>2</sub>. Especially, the Pd-loaded SnO<sub>2</sub> sensor possesses good sensitivity (0.58 ppm<sup>-1</sup>) at low concentration (0.05–1 ppm) with excellent linearity (R<sup>2</sup> = 0.9963) and low detection limit (50 ppb). The high sensing performance of Pd-loaded SnO<sub>2</sub> are attributed to the excellent adsorption and catalysis effects of Pd nanoparticle. Meaningfully, the potential applications of C<sub>2</sub>H<sub>4</sub> sensor are performed for monitoring the maturity and freshness of fruits, which presents a promising prospect in fruit quality evaluation.

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Ethylene (C<sub>2</sub>H<sub>4</sub>) as a colorless and odorless phytohormone is closely related to the maturity and freshness of fruit [1]. For example, low concentration of C<sub>2</sub>H<sub>4</sub> (from 10 ppb to 10 ppm) would be released from immature fruit, and C<sub>2</sub>H<sub>4</sub> also triggers ripening genes resulting in changes in texture, color and taste of fruit [2]. With the fruit maturing, the C<sub>2</sub>H<sub>4</sub> production increases gradually to over 100 ppm. Thus, the emission of C<sub>2</sub>H<sub>4</sub> can be used as an indicator of fruit maturity [3]. Furthermore, excess C<sub>2</sub>H<sub>4</sub> during fruit storage can accelerate respiratory rates of fruits resulting in senescence and even spoilage [4]. Therefore, in order to evaluate the quality of fruit efficiently and intelligently, it is important to develop a high performance gas sensor for detecting C<sub>2</sub>H<sub>4</sub> concentration at trace level.

Up to now, various techniques, including optical, electrical and mass-sensitive methods, have been applied for C<sub>2</sub>H<sub>4</sub> detection. Among them, gas chromatography [5], fluorescence [6], photo-acoustic spectroscopy [7] could achieve high-sensitive detection of

C<sub>2</sub>H<sub>4</sub>, but the expensive cost, bulky and non-portability of these methods largely impede their application in fruit detection. Instead, the resistive-type sensor are worth exploring because of its simple and affordable. Due to the limited physicochemical reactivity of C<sub>2</sub>H<sub>4</sub>, C<sub>2</sub>H<sub>4</sub> sensors working at room temperature have poor performance [8–11]. In order to accelerate the adsorption and desorption of nonpolar C<sub>2</sub>H<sub>4</sub> molecules on sensitive materials, it is necessary to use assistant methods such as heating [12] or ultraviolet irradiation [13]. Several C<sub>2</sub>H<sub>4</sub> sensors based on heating have been reported in the literature [14–20]. For example, Kathirvelan *et al.* studied the C<sub>2</sub>H<sub>4</sub> sensing properties of titanium dioxide and tungsten trioxide (TiO<sub>2</sub>-WO<sub>3</sub>) nanocomposites at 250 °C, the minimum achievable concentration was 8 ppm for continuous detection and the response time is unknown [14]. Nimitrakoolchai *et al.* reported a WO<sub>3</sub> gas sensor prepared by precipitation method, the detection range is 3–10 ppm and the response time is about a few minutes for 3–5 ppm C<sub>2</sub>H<sub>4</sub> [15]. Wang *et al.* employed a porous zinc oxide (ZnO) nanosheets based C<sub>2</sub>H<sub>4</sub> gas sensor, it is of 5 ppm detection limit and the response less than 2 at 500 °C [16]. Progress has been made in resistive-type C<sub>2</sub>H<sub>4</sub> sensors, but the detection limit, response value and response time still need to be improved.

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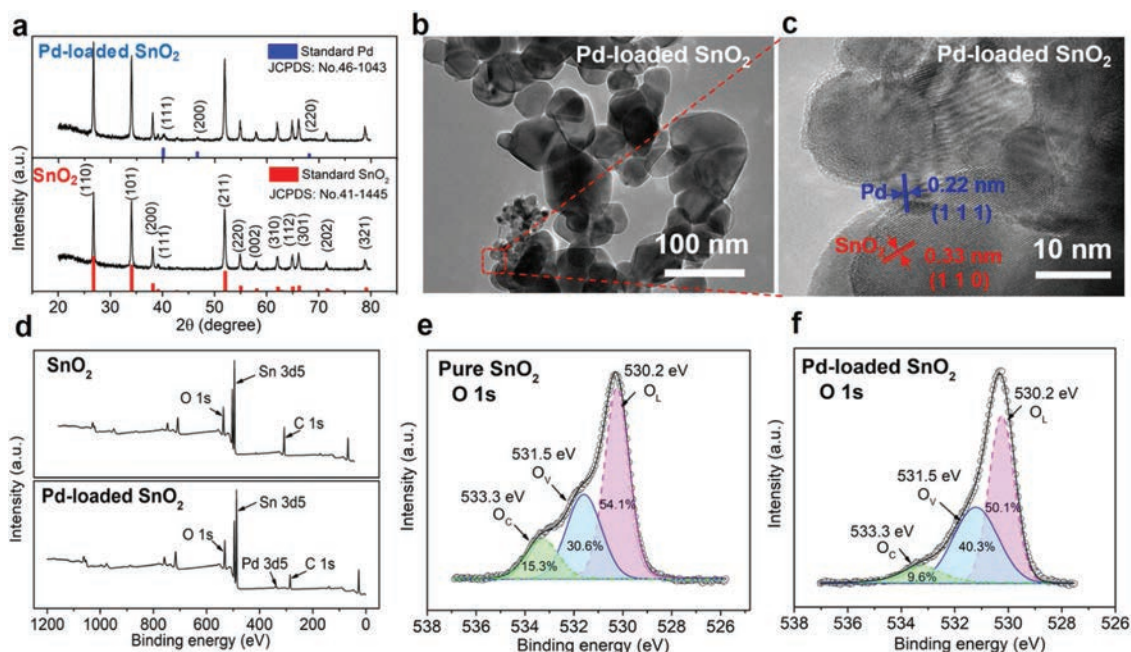
In order to enhance the performance of gas sensor, using noble metals (e.g., Pt [17], Au [21], Ag [22] and Pd [23]) loaded metal oxide semiconductors (MOS) is a prevalent strategy. In particular, as a potential  $C_2H_4$  acceptor and catalyst [24–26], Pd loaded MOS is expected to achieve high response, fast speed and low detection limit. Herein, Pd-loaded tin oxide ( $SnO_2$ ) is used to prepare  $C_2H_4$  gas sensor. The  $C_2H_4$  sensing properties are investigated at 50–400 °C, and the  $C_2H_4$  gas sensing mechanism is discussed in this paper. Particularly, the potential application of Pd-loaded  $SnO_2$   $C_2H_4$  sensor has been developed in fruit detection.

The Pd-loaded  $SnO_2$   $C_2H_4$  sensor (Fig. S1a in Supporting information) was fabricated by simple coating method, and the details referred to the supplementary material. The micro-morphologies, lattice images, crystallization and chemical valence states of pristine  $SnO_2$  and Pd-loaded  $SnO_2$  were characterized by various techniques including scanning electron microscopy (SEM, Zeiss, SUPRA 55), transmission electron microscope (TEM, FEI, Tecnai G2 F20 S-TWIN), X-ray diffraction (XRD, Bruker, D8 Advance) with a range of  $2\theta$  from 20° to 80° at 40 kV and 40 mA using Co-K $\alpha$  as the irradiation source ( $\lambda = 0.1789$  nm), and X-ray photoelectron spectroscopy (XPS, Thermo Fisher Scientific, K-Alpha), respectively.

The  $C_2H_4$  sensing performance of sensors were measured by a dynamic homemade system with the Keithley 2700 data acquisition instrument and a regulated power supply (Fig. S1b in Supporting information). The operating temperature of sensor was varied between 50 °C and 400 °C. In order to simulate the conventional atmospheric environment, the testing relative humidity (RH) is controlled at 51.9% by air with bubbling distilled water [27,28], and calibrated by a high-accuracy hygrometer (CEM, DT-625, Shenzhen Everbest Machinery Industry Co., Ltd., China). The response of gas sensor is calculated as following equation:  $Response = R_a/R_g$  ( $R_a$ : resistance under the air,  $R_g$ : resistance under the target gas). The sensitivity is defined as the slope of response–concentration linear fitting curve. And the response/recovery times are defined as the time taken by  $C_2H_4$  gas sensor to achieve 90% of the total resistance change in the case of adsorption and desorption respectively.

The SEM images of pristine  $SnO_2$  and Pd-loaded  $SnO_2$  are shown in Fig. S2 (Supporting information). It can be seen that both of them are composed of nanoparticles, but there is no obvious difference in microscopic morphology of two SEM images. In order to confirm the existence of Pd, the XRD patterns of pristine  $SnO_2$  and Pd-loaded  $SnO_2$  are characterized in Fig. 1a. All the diffraction peaks of  $SnO_2$  can be well coincided with JCPDS card No. 41-1445 and match the tetragonal rutile structure [29], the diffraction peaks of Pd are tally with JCPDS data No. 46-1043, which reflect the face-centered cubic crystalline structure [30]. Compared with the prominent peaks of  $SnO_2$ , the weak diffraction peaks of Pd indicate that the trace existence of Pd in composite film. The TEM images of Pd-loaded  $SnO_2$  are performed as shown in Figs. 1b and c. The lattice fringes with an inter-planar distance of 0.33 nm coincide well with the (110) plane of  $SnO_2$  in Fig. 1c. And the lattice distance of 0.22 nm is correspond to the (111) crystalline plane of Pd, which further suggests that the Pd-loaded  $SnO_2$  nanoparticles have been successfully synthesized from the micro-morphology. It is beneficial to regulate the interface barrier between  $SnO_2$  and Pd [31].

The surface compositions and chemical valence states are analyzed by XPS. Fig. 1d is the full range XPS spectra of pristine  $SnO_2$  and Pd-loaded  $SnO_2$ . Fig. 1e displays the O 1s spectra of  $SnO_2$  and the binding energies at 530.2, 531.5 and 533.3 eV can be assigned to lattice oxygen species ( $O_L$ ), oxygen vacancy ( $O_V$ ) and the chemisorbed oxygen species ( $O_C$ ,  $H_2O$  species and impurity oxygen) on the surface of sensing materials, respectively [32–34]. The oxygen vacancy has significant influence on the gas sensing properties of resistance-type MOS [33]. Compared Fig. 1f with Fig. 1e, the content of oxygen vacancy in Pd-loaded  $SnO_2$  (40.3%) is higher than pristine  $SnO_2$  (30.6%). The increase of oxygen vacancies of Pd-loaded  $SnO_2$  is conducive to  $C_2H_4$  reaction and further enhance the sensing performance [34]. The higher content of chemisorbed oxygen (15.3%) in pristine  $SnO_2$  could be due to the adsorption of water molecules [35,36]. The high-resolution XPS spectra of Pd 3d is shown in Fig. S3 (Supporting information), the Pd 3d<sub>5/2</sub> electron binding energies in palladium oxide (PdO) and metallic Pd are 335.9 and 334.5 eV, respectively [31]. The contents of PdO of Pd-loaded  $SnO_2$  are calculated to be 58%, it is confirmed



**Fig. 1.** (a) XRD patterns of pristine  $SnO_2$  and Pd-loaded  $SnO_2$ . (b) TEM image of Pd-loaded  $SnO_2$ . (c) HRTEM image of Pd-loaded  $SnO_2$ . (d) XPS fully scanned spectra. XPS spectra: O 1s spectra of (e) pristine  $SnO_2$  and (f) Pd-loaded  $SnO_2$ .

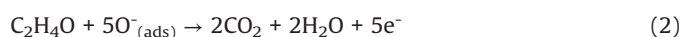
that Pd nanoparticles are easy to oxidize into PdO in air due to their strong electron transport ability. Moreover, the conversion of Pd and PdO in the oxygen environment plays an important role in MOS sensitization [24].

The working temperature has great influence on the response of MOS based gas sensors [37]. In order to investigate the relationship between operating temperature and the C<sub>2</sub>H<sub>4</sub> sensing properties, the response of sensors based on pristine SnO<sub>2</sub> and Pd-loaded SnO<sub>2</sub> to 10 ppm C<sub>2</sub>H<sub>4</sub> at operating temperatures ranging from 50 °C to 400 °C are shown in Fig. 2a. As can be seen, the addition of Pd reduces the optimum working temperature of gas sensor from 350 °C to 250 °C. The phenomena of the increasing sensing response and the decreasing working temperature for SnO<sub>2</sub> are consistent with the previous researches by Zhang's group [38], which directly verifies the promotion effect of Pd nanoparticles. Fig. 2b displays the real-time resistance variation curves of two gas sensors to 20–100 ppm C<sub>2</sub>H<sub>4</sub> at 250 °C. When exposed to 100 ppm C<sub>2</sub>H<sub>4</sub>, the response of Pd-loaded SnO<sub>2</sub> sensor (11.1) is about 3 times higher than that of pristine SnO<sub>2</sub> (3.5). Moreover, the response time is also significantly shortened from 7 s to 1 s compared with pristine SnO<sub>2</sub>. As shown in Fig. 2c, both of two sensors exhibit good linearity, but the Pd-loaded SnO<sub>2</sub> gas sensor holds a larger sensitivity (0.05 ppm<sup>-1</sup>) than that of pristine SnO<sub>2</sub> one (0.02 ppm<sup>-1</sup>). Fig. 2d displays the linear fitting curves of two sensors to low concentration C<sub>2</sub>H<sub>4</sub> (0.05–1 ppm), the Pd-loaded SnO<sub>2</sub> gas sensor exhibits better sensitivity (0.58 ppm<sup>-1</sup>) with excellent linearity (R<sup>2</sup> = 0.9963). Especially, the Pd-loaded SnO<sub>2</sub> gas sensor achieves ultra-low detection concentration (50 ppb) (the inset of Fig. 2d). The real-time resistance variation curves of two sensors corresponding to Fig. 2d are shown in Figs. S4 and S5 (Supporting information). Fig. 2e exhibits the response and recovery curves for three cycles when the Pd-loaded SnO<sub>2</sub> gas sensor is exposed to 100 ppb, 1 ppm and 40 ppm C<sub>2</sub>H<sub>4</sub> concentrations. As can be seen, the response and recovery curves can switch to the steady states after the C<sub>2</sub>H<sub>4</sub> level changed, indicating that the sensor shows good repeatability in the continuous measurements. Fig. 2f presents the good stability and durability of the Pd-loaded SnO<sub>2</sub> gas sensor for low concentration

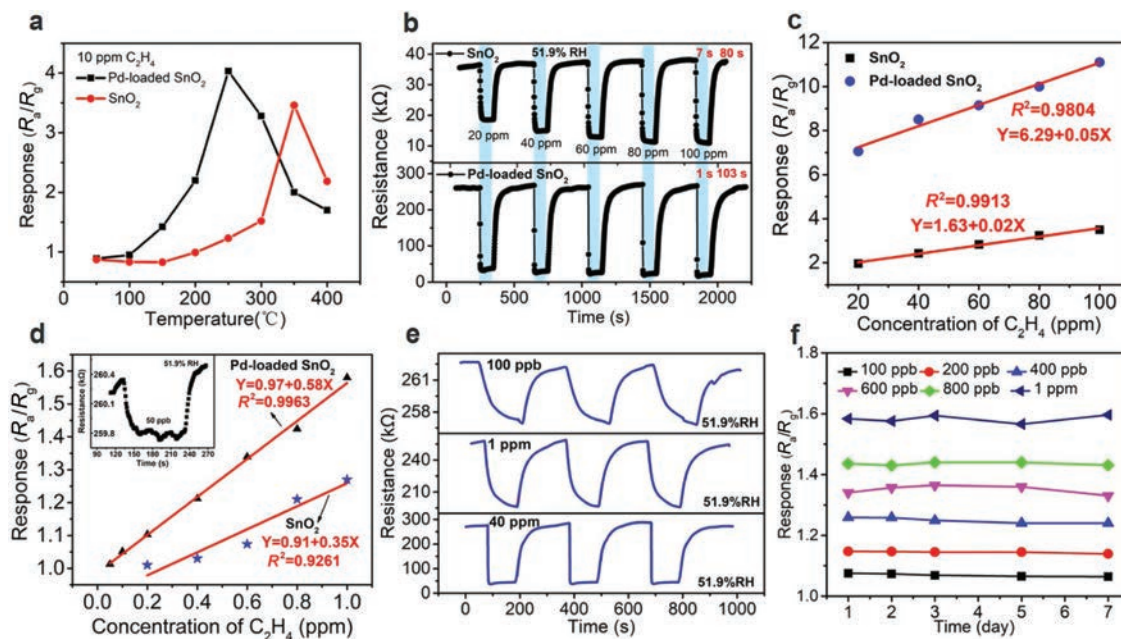
C<sub>2</sub>H<sub>4</sub> in a week. Moreover, the other sensing performance (humidity, selectivity) of Pd-loaded SnO<sub>2</sub> gas sensor are studied in supplementary material. Compared with previous works [8–11,14–20], the C<sub>2</sub>H<sub>4</sub> sensor based on Pd-loaded SnO<sub>2</sub> shows high performance in terms of the response value (R<sub>a</sub>/R<sub>g</sub> = 11.1 for 100 ppm), detection limit (≤50 ppb) and response speed (1 s for 100 ppm), which are listed detailedly in Table 1.

A plausible explanation of C<sub>2</sub>H<sub>4</sub> sensing mechanism for Pd-loaded SnO<sub>2</sub> sensor is discussed as follows. As shown in Fig. 3a, since the work function of Pd (5.12 eV) is higher than that of SnO<sub>2</sub> (4.5 eV), the Schottky junction will be formed at the interface of Pd and SnO<sub>2</sub>, which causes the part of electrons transfer from SnO<sub>2</sub> to Pd [39]. It is beneficial to the formation of depletion region around SnO<sub>2</sub>. When the Pd-loaded SnO<sub>2</sub> gas sensor is exposed to air (Fig. 3b), oxygen molecules combine with electrons to form adsorbed oxygen (O<sup>-</sup> at 250 °C) on the surface area of Pd-loaded SnO<sub>2</sub>, resulting in a thick electron-depleted layer [40]. Furthermore, Pd (PdO) as a catalyst activates the dissociation of oxygen molecules and enhances the adsorption activity of oxygen on the surface of SnO<sub>2</sub> [41]. The increase of oxygen vacancy of Pd-loaded SnO<sub>2</sub> is also proved by XPS analysis. The thicker electron-depleted layer results in the increase of air resistance of SnO<sub>2</sub> after Pd loading [24], which is beneficial to improve the C<sub>2</sub>H<sub>4</sub> sensing response of SnO<sub>2</sub> sensor.

When the Pd-loaded SnO<sub>2</sub> gas sensor is exposed to C<sub>2</sub>H<sub>4</sub> gas (Fig. 3c), C<sub>2</sub>H<sub>4</sub> molecules with a carbon-carbon double bond (C=C) are oxidized to ethylene oxide or acetaldehyde, then further reacted to produce carbon dioxide (CO<sub>2</sub>) and water (H<sub>2</sub>O), and the reaction process can be shown as follows [42]:



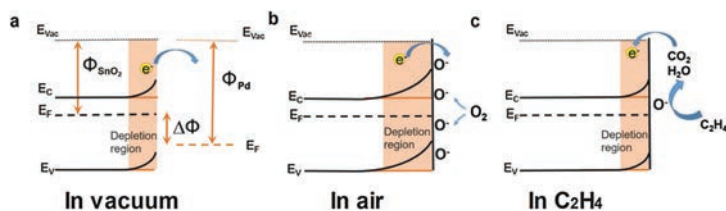
The process of C<sub>2</sub>H<sub>4</sub> oxidation releases the trapped electrons back to the conduction band, leading to the reduction of depletion region and the decrease in resistance. As the potential receptors of



**Fig. 2.** (a) Response of the pristine SnO<sub>2</sub> and Pd-loaded SnO<sub>2</sub> sensors to 10 ppm C<sub>2</sub>H<sub>4</sub> at different working temperature. (b) Real-time resistance variation curves and (c) linear fitting curves of response for pristine SnO<sub>2</sub> and Pd-loaded SnO<sub>2</sub> sensors to 20–100 ppm C<sub>2</sub>H<sub>4</sub> at 250 °C. (d) Linear fitting curves of response to 0.05–1 ppm C<sub>2</sub>H<sub>4</sub> at 250 °C, the inset of (d) is the resistance variation curves to 50 ppb C<sub>2</sub>H<sub>4</sub>. (e) Repeatability curves. (f) Stability curves in a week.

**Table 1**  
Gas sensing properties of reported resistive-type C<sub>2</sub>H<sub>4</sub> gas sensors.

Materials	Response ( $R_a/R_g$ )	Detection limit	Response time (s)	Temperature (°C)	References
SWNTs/copper(I) complex	<2 (50 ppm)	500 ppb	–	r.t.	[8]
PNAI-H <sub>2</sub> SO <sub>4</sub>	1.25 (100 ppm)	10 ppm	–	r.t.	[9]
MWCNTs	< 2 (100 ppm)	10 ppm	10	r.t.	[10]
PANI/MWCNTs/SnO <sub>2</sub>	~1.2 (100 ppm)	10 ppm	–	r.t.	[11]
TiO <sub>2</sub> -WO <sub>3</sub>	~1.25 (100 ppm)	8 ppm	–	250	[14]
WO <sub>3</sub>	1.1 (3 ppm)	3 ppm	>60	300	[15]
Porous ZnO	< 2 (100 ppm)	5 ppm	8	500	[16]
SnO <sub>2</sub> -1% Pt	3.16 (100 ppm)	1 ppm	–	250	[17]
Silicalite-1/ SnO <sub>2</sub>	~3.9 (70 ppm)	8 ppm	14	350	[18]
SnO <sub>2</sub>	10.43 (100 ppm)	10 ppm	270	300	[19]
WO <sub>3</sub>	~2 (200 ppm)	200 ppm	~67	350	[20]
SnO <sub>2</sub>	5 (200 ppm)	200 ppm	~83	350	[20]
Pd-SnO <sub>2</sub>	11.1 (100 ppm)	50 ppb	1	250	This work



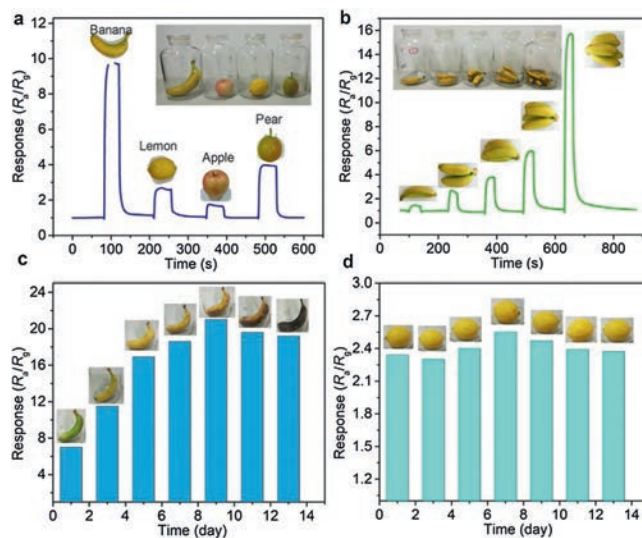
**Fig. 3.** Schematic diagram of C<sub>2</sub>H<sub>4</sub> sensing mechanism for Pd-loaded SnO<sub>2</sub> gas sensor. Schottky barrier in (a) vacuum, (b) air and (c) C<sub>2</sub>H<sub>4</sub>.

C<sub>2</sub>H<sub>4</sub>, Pd nanoparticles can increase the number of efficient adsorption sites of C<sub>2</sub>H<sub>4</sub> molecules near the surface of sensitive film [26]. Moreover, adsorbed C<sub>2</sub>H<sub>4</sub> molecules on Pd will spillover onto the surface of SnO<sub>2</sub> because of the spillover effect [43], leading to the high sensing response. In summary, the C<sub>2</sub>H<sub>4</sub> gas-sensing mechanism of Pd-loaded SnO<sub>2</sub> can be explained by depletion layer theory, the catalysis and receptor function of Pd.

The C<sub>2</sub>H<sub>4</sub> emission can be used as an indicator to estimate the coloration, texture, and flavor taste of fruits [44]. Inspired by the demand for fruit freshness and ripeness monitoring, potential applications of Pd-loaded SnO<sub>2</sub> gas sensor are explored. The temperature and humidity of the laboratory are controlled at 25 ± 2 °C and 50% ± 5% RH by an air conditioner, and recorded by the high-accuracy hygrometer. Fig. 4 displays the response curves in the application of fruit quality detection. In order to simulate the storage conditions and observe morphological changes, fruits are put into a transparent bottle packed with porous plastic wrap to avoid fermentation, and then place the gas sensor at the bottle mouth during detection. Fig. 4a shows the real-time response curves of the gas sensor to four fresh fruits (banana, lemon, apple and pear), the weight of each fruit is about 120–150 g. It can be seen that the Pd-loaded SnO<sub>2</sub> gas sensor presents different responses to different fruits. Fig. 4b displays the increasing response to gradually increasing banana weight (about 25 g, 50 g, 75 g, 100 g and 125 g). The nonlinear increase indicates that C<sub>2</sub>H<sub>4</sub> emissions are mutually reinforcing [2]. To further support potential applications in fruit storage, Fig. 4c presents the variation in response to one banana at different ripening stages. As a comparison, the same test to lemon is shown in Fig. 4d. The results show that the response of banana changed significantly from non-climacteric to climacteric, while that of lemon changed smoothly. As a respiratory climacteric fruit, banana is susceptible to the influence of C<sub>2</sub>H<sub>4</sub> concentration in senescence and metamorphosis, while the non-respiration climacteric lemon is more resistant to storage than banana due to the slow metabolism of lemon, which is consistent with the experimental results. In short, the above tests demonstrate that different species, quantities and storage states of fruits can be estimated by the Pd-loaded

SnO<sub>2</sub> gas sensor like a smart nose, which suggest that the potential applications of Pd-loaded SnO<sub>2</sub> gas sensor in fruits quality evaluation.

In summary, a Pd-loaded SnO<sub>2</sub> C<sub>2</sub>H<sub>4</sub> sensor for fruit quality evaluation is fabricated by coating method. The high sensing performance of Pd-loaded SnO<sub>2</sub> C<sub>2</sub>H<sub>4</sub> sensor are obtained at 250 °C, it is of high response (11.1 for 100 ppm), fast speed (1 s for 100 ppm), good linearity ( $R^2 = 0.9804$  for 20–100 ppm), low detection limit (50 ppb) and good stability. The sensing mechanism of Pd-loaded SnO<sub>2</sub> sensor is discussed by the depletion layer theory, the catalysis and receptor function of Pd. Furthermore, the applications of the Pd-loaded SnO<sub>2</sub> sensor in detecting fruit species, quantities and storage states are explored. The results show that



**Fig. 4.** Potential applications of the Pd-loaded SnO<sub>2</sub> gas sensor: real-time response curves to (a) various fruits, (b) bananas with different quantities, (c) one banana in half a month, (d) one lemon in half a month.

the prepared Pd-loaded SnO<sub>2</sub> sensor has great potential for developing high-performance C<sub>2</sub>H<sub>4</sub> sensor in fruit quality monitoring.

### Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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### Appendix A. Supplementary data

Supplementary material related to this article can be found, in the online version, at doi:<https://doi.org/10.1016/j.ccl.2020.04.032>.

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